

(12) INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(19) World Intellectual Property Organization  
International Bureau



(43) International Publication Date  
6 September 2002 (06.09.2002)

PCT

(10) International Publication Number  
**WO 02/068485 A1**

(51) International Patent Classification<sup>7</sup>: **C08F 290/02**,  
299/00, 293/00

(21) International Application Number: PCT/US01/49523

(22) International Filing Date:  
21 December 2001 (21.12.2001)

(25) Filing Language: English

(26) Publication Language: English

(30) Priority Data:  
60/257,738 22 December 2000 (22.12.2000) US

(71) Applicant: **CARNEGIE MELLON UNIVERSITY**  
[US/US]; 5000 Forbes Avenue, Pittsburgh, PA 15213  
(US).

(72) Inventors: **MATYJASZEWSKI, Krzysztof**; 9 Queens  
Court, Pittsburgh, PA 15238 (US). **LUTZ, Jean-Francois**;  
Apt. 3B, 6204 5th Avenue, Pittsburgh, PA 15238 (US).  
**SHINODA, Hosei**; \* (JP).

(74) Agents: **PIKE, Bernard, G.** et al.; Kirkpatrick & Lock-  
hart LLP, Henry W. Oliver Building, 535 Smithfield Street,  
Pittsburgh, PA 15222-2312 (US).

(81) Designated States (*national*): AE, AG, AL, AM, AT, AU,  
AZ, BA, BB, BG, BR, BY, BZ, CA, CH, CN, CO, CR, CU,  
CZ, DE, DK, DM, DZ, EC, EE, ES, FI, GB, GD, GE, GH,  
GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC,  
LK, LR, LS, LT, LU, LV, MA, MD, MG, MK, MN, MW,  
MX, MZ, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK,  
SL, TJ, TM, TR, TT, TZ, UA, UG, UZ, VN, YU, ZA, ZW.

(84) Designated States (*regional*): ARIPO patent (GH, GM,  
KE, LS, MW, MZ, SD, SL, SZ, TZ, UG, ZM, ZW),  
Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM),  
European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR,  
GB, GR, IE, IT, LU, MC, NL, PT, SE, TR), OAPI patent  
(BF, BJ, CF, CG, CI, CM, GA, GN, GQ, GW, ML, MR,  
NE, SN, TD, TG).

**Published:**

- with international search report
- before the expiration of the time limit for amending the  
claims and to be republished in the event of receipt of  
amendments

*For two-letter codes and other abbreviations, refer to the "Guid-  
ance Notes on Codes and Abbreviations" appearing at the begin-  
ning of each regular issue of the PCT Gazette.*

(54) Title: PROCESS FOR PREPARATION OF GRAFT POLYMERS

(57) Abstract: A polymerisation process comprising the polymerisation of macromers and eventually comonomers with a com-  
patible macroinitiator to form a graft (co)polymer.



**WO 02/068485 A1**

## TITLE OF INVENTION

## Process for Preparation of Graft Polymers

5

## FIELD OF THE INVENTION

This invention is directed toward a process for the preparation of graft (co)polymers. Specifically, graft (co)polymers prepared by "grafting through" macromonomers with a macroinitiator. The disclosed process is applicable to any (co)polymerization process. Embodiments of the process of the present invention provides for the preparation of graft copolymers with a broad range of structures and compositions including, but not limited to, phase separable compositions and compositions with controlled distribution of the graft copolymers along the backbone polymer and graft copolymers comprising biologically compatible grafts or grafts comprising polyolefins. The invention also allows for preparation of block-graft copolymers wherein each block is a graft copolymer.

## DESCRIPTION

The properties of graft (co)polymers are determined by many structural and chemical factors including, but not limited to, the chemical composition of the backbone of the (co)polymer, the chemical composition of the branch (co)polymers, the length of the backbone, the lengths of the branches, and the spacing of the branches along the backbone or the branch distribution. The "grafting through" polymerization process is considered potentially one of the most useful ways to design and obtain well-defined graft (co)polymers. The "grafting through" process for the preparation of graft (co)polymers comprises polymerizing macromonomers. The macromonomers may preferably be copolymerized with a lower molecular weight comonomer to prepare a graft (co)polymer. Schulz, G. O.; Milkovich, R. *J. Appl. Polym. Sci.* 1982, 27, 4773.

Graft (co)polymers have many commercially important applications. Many specific applications of graft (co)polymers benefit from the phase separation of the graft (co)polymer between the backbone of the (co)polymer and the grafts

of the (co)polymer. This phase separation of the polymer often results from a difference in phylicities between the grafts and the backbone. However, the benefit derived from the phylicity differences in the graft copolymer can also create difficulty incorporating the macromonomers into polymer. Some of these  
5 difficulties may be attributed to the slow rate of diffusion of the polymerizable functional group on the macromonomer to the growing polymer chain end.

However, phase separation due to the differences of phylicities of the components of the polymerization medium also contributes to the copolymerization processing problems. The phase separation is particularly significant in the preparation of  
10 high performance graft copolymers, such as organic/inorganic hybrid graft copolymers. These high performance polymers can comprise dramatically different characteristics in the backbone and in the branches. The effect of phase separation during polymerization may be reduced in some cases by using a compatible co-solvent for both the macromonomer and the copolymer. However,  
15 if the macromonomer and the copolymer segment are highly incompatible, the co-solvent may be not sufficient to decrease the incompatibility effect.

The polymerization medium of a radical polymerization process comprising macromonomers and monomers, for example, may undergo a phase separation early in the polymerization process as the backbone grows from the  
20 polymerization of the monomer, a high molecular weight copolymer is produced. The phylicity of this high molecular weight copolymer may be different than the phylicity of the macromonomer. This difference in phylicities may lead to a two phase polymerization process, one phase comprising a substantial portion of the growing polymer chain and the low molecular weight monomer and the other  
25 phase comprising a substantial portion of the macromonomer.

This phenomenon leads to a heterogenous distribution of grafts among the chains, some polymer having many grafts and some polymer having few grafts. The amount of grafts on a polymer depend on whether the polymer was formed early in the polymerization and later in the polymerization.

30 Prior art processes for the preparation of macromonomers have been limited primarily to ionic polymerization processes and this has limited the functionality that can be ultimately incorporated into polymers and thereby the

control over factors effecting the properties of the graft copolymer. Attempts have been made to reduce the phase separation problem by addition of an appropriate solvent to the polymerization process but commercial processes are still severely limited in the range of compositions of the graft (co)polymers that can be prepared.

### DESCRIPTION OF THE FIGURES

The features and advantages of embodiments of the present invention may be better understood by reference to the accompanying figures, in which:

Figure 1 is a graphical depiction of graft copolymers, Polymer A is an example of a graft copolymer having homogeneously distributed grafts and Polymer B is an example of a graft copolymer having a heterogeneous distribution of grafts;

Figure 2 re graphs of the kinetic data of the copolymerization and Jaacks plot for the ATRP copolymerization of MMA and PDMS macromonomer in xylene solution at 90 °C (HS166), the data indicates slow consistent incorporation of the macromonomer into the copolymer;

Figure 3 is a graphical representation of examples of different structures of the block graft copolymers comprising PDLA and PDMS branches and PMMA backbone that may be produced by the process of the present invention;

Figure 4 is a graphical representation of an example of the structure of the synthesized gradient graft copolymer comprising a gradient in the concentration of the concentration of the two grafts along the length of the backbone.

### DESCRIPTION OF THE EMBODIMENTS OF THE INVENTION

The present invention provides a polymerization process for the preparation of graft (co)polymers by the grafting through process. An embodiment of the polymerization process of the present invention comprises copolymerizing macromonomers with (co)monomers utilizing a macroinitiator to form a graft (co)polymer. Another embodiment of a polymerization process of the present invention comprises (co)polymerizing macromonomers and monomers with a graft



copolymer macroinitiator to form a block-graft (co)polymer. As used herein, a polymerization process is a system comprising the components necessary to conduct a any polymerization, such as for example, in an embodiment of a standard free radical or addition polymerization system the polymerization process comprises radically polymerizable monomers and a standard free radical initiator, or in an embodiment of a nitroxide mediated polymerization comprises an initiator, a stable free radical and radically polymerizable monomers, etc. Another embodiment of the process of the present invention comprises (co)polymerizing macromonomers and monomers with a compatible macroinitiator.

As used herein, a "macroinitiator" is an oligomer, polymer or other large molecule which is capable of initiating a polymerization process. A macroinitiator may comprise more than one polymerization initiation site. A "compatible macroinitiator" is a macroinitiator which is at least partially soluble or miscible in the macromonomer and does not undergo a phase separation. The compatible macroinitiator may be present in a system with mixtures of macromonomers or mixtures of macromonomers and other monomers. A compatible macroinitiator, if necessary, acts to compatibilize the macromonomer and at least one or a combination of the monomer, the initially produced polymer. The compatible macroinitiator may have the same segment structure as the macromonomer, or more preferably, a macroinitiator that has a different composition than the macromonomer but is soluble in macromonomer thereby using entropy as an additional driving force for compatibility between the polymerization components and the forming copolymer, all in solution in remaining (co)monomer(s). Improved reactivity for the macromonomer can be expected because the growing copolymer chain is a block copolymer comprising the (co)polymerization components and the first formed copolymer acts as a surfactant and retains compatibility with the macromonomer.

As used herein, "macromonomers" are oligomers, polymers or other large molecules comprising at least one functional group suitable for further (co)polymerization, preferably the functional group is a terminal functional group. One such terminal functional group is an unsaturated functional group suitable for free radical (co)polymerization. A "radically copolymerizable macromonomer" is a

macromonomer which is capable of reacting in a radical polymerization process, preferably a radically polymerizable macromonomer comprises a terminal unsaturated functional group capable of reacting with a radical.

In addition, unless otherwise indicated, all numbers expressing quantities of concentrations, time, temperatures, and so forth used in the present specification and claims are to be understood as being modified in all instances by the term "about." Accordingly, unless indicated to the contrary, the numerical parameters set forth in the following specification and claims are approximations that may vary depending upon the desired properties sought to be obtained by the present invention. At the very least, and not as an attempt to limit the application of the doctrine of equivalents to the scope of the claims, each numerical parameter should at least be construed in light of the number of reported significant digits and by applying ordinary rounding techniques.

Notwithstanding that the numerical ranges and parameters setting forth the broad scope of the invention are approximations, the numerical values set forth in the specific examples are reported as precisely as possible. Any numerical value, however, may inherently contain certain errors necessarily resulting from the standard deviation found in their respective testing measurements.

An embodiment of the process of the present invention comprises macromonomers and compatible macroinitiators in a controlled radical polymerization process for the preparation graft copolymers by "grafting through" polymerization. This embodiment allows incorporation of macromonomers into a growing backbone polymer and controlled growth of the backbone polymer. The chemical and structural properties of the product graft (co)polymer may be controlled by use of a compatible macroinitiator and the functional group on the macromonomer which effect the relative rates of incorporation of the macromonomer and the monomer.

In a further embodiment of the present invention, a copolymer with well-defined branches or grafts with predetermined molecular weights and low polydispersities can be obtained by preparing macromonomers by a living or a living/controlled polymerization process. The meaning of these terms are

discussed in the following references: Szwarc, M. *Nature* 1956, 178, 1168-1169; Matyjaszewski, K. *J. Phys. Org. Chem.* 1995, 8, 197-207; Matyjaszewski, K.; Lin, C. H. *Makromol. Chem., Macromol. Symp.* 1991, 47, 221-237; Litvinenko, G.; Mueller, A. H. E. *Macromolecules* 1997, 30, 1253-1266; Matyjaszewski, K.;

5 Mueller, A. H. E. *Polym. Prepr.* 1997, 38(1), 6-9 and Matyjaszewski, K. *J. Macromol. Sci., Pure Appl. Chem.* 1994, A31, 989-1000. In a preferred embodiment of the present invention the backbone is also polymerized with using a controlled polymerization process of the macromonomer and the low molecular weight comonomer with a macroinitiator thereby producing a backbone copolymer  
10 with a controlled molecular weight distribution. As used herein, "monomer" or "comonomer" is a molecule that is capable of conversion to polymers, synthetic resins or elastomers by combination with itself or other similar molecules or compounds. The term monomer, as used herein, is not limited to small molecules, but includes oligomers, polymers and other large molecules capable of  
15 combining with themselves or other similar molecules or compounds.

The reactivity ratio of the macromonomer with the comonomer has been used to measure incorporation of macromonomers into a graft (co)polymer. A reactivity ratio close to 1 indicates a more statistical incorporation of the macromonomer into the graft (co)polymer and would result in the preparation of a  
20 graft (co)polymer with a homogeneous distribution of grafts along the backbone (co)polymer. Reactivity ratios different than 1 indicate either a more rapid incorporation or a less rapid incorporation of the macromonomer or the comonomer and result in a more heterogeneous distribution of grafts along the backbone of the (co)polymer.

25 One embodiment of the process of the present invention, such as embodiments comprises copolymerizing a macromonomer with a vinyl-end group and a radically (co)polymerizable monomer in a controlled radical polymerization process, such as atom transfer radical polymerization ("ATRP"), is described in detail. This embodiment exemplifies the criteria for grafting through  
30 macromonomers into graft (co)polymers, not only for controlled radical polymerization processes but also for any other polymerization process, such as nitroxide mediated polymerization, ionic or cationic polymerization, RAFT

processes and addition polymerization processes. Further embodiments of the process of the present invention include the preparation of hybrid graft (co)polymers with grafts or backbones comprising inorganic monomer units, grafts or backbones comprising biodegradable monomer units, grafts or backbones comprising polyolefin monomer units and the preparation of block (co)polymers wherein each block comprises a graft (co)polymer, and preferably each block comprises graft segments of different compositions. Also described are embodiments which can be utilized to prepare gradient-graft copolymers, gradient-graft-(co)macroinitiators and gradient-graft-(co)macromonomers. ATRP processes are disclosed in the commonly assigned U.S. Patents 5,763,546; 5,807,937; 5,789,487; 5,945,491; 6,111,022; 6,121,371; 6,124,411; 6,162,882; and U.S. Patent Applications 09/034,187; 09/018,554; 09/431,871; 09/359,359; 09/359,591; 09/369,157; 09/534,827; 09/972,046; 09/872,056; and 09/972,260 all of which are herein incorporated by reference.

Embodiments of the process of the present invention may also be used for the preparation of well-defined homogeneous graft (co)polymers, biphasic graft copolymers and triphasic graft copolymers. A biphasic copolymer is a polymer which contains segments which form into two separate phases either alone or in at least one solvent, such as, for example, a segment soluble in an organic phase and a segment soluble in an aqueous phase. A triphasic copolymer is a polymer which contains segments which form into three separate phases either alone or in at least one solvent, such as, for example, a segment soluble in an organic phase, a segment soluble in an aqueous phase and a crystalline phase or a low T<sub>g</sub> segment, a high T<sub>g</sub> segment and a crystalline phase.

The properties of graft (co)polymers are determined by many structural and chemical factors including, but not limited to, the chemical composition of the backbone of the (co)polymer, the chemical composition of the branch (co)polymers, the length of the backbone, the lengths of the branches, and spacing of the branches along the backbone or the branch distribution. The average spacing of the side chains, and the overall (co)polymer composition, will be determined by the molar ratio of macromonomer to comonomer in the feed, their reactivity ratios and the ratio of comonomers incorporated into graft

copolymer. For example, in Figure 1, both Polymer A and Polymer B have the similar backbone lengths, similar branch lengths, and the similar number of branches, but clearly a different branch spacing distribution. Polymer A has homogeneously distributed branch spacing, while polymer B has heterogeneously distributed branch spacing structures. As used herein, "homogeneously distributed branch spacing" means that the branches are distributed generally evenly throughout the backbone. In a "heterogeneously distributed branch spacing," the (co)polymer has regions with higher concentrations of branches. The process of the present invention may be used to control the spacing distribution of the branches or grafts along the polymer backbone with the modification of the components or parameters of the polymerization, such as, but not limited to, selection of the macroinitiator, the reactive functional group on the macromonomer, polymerizing a mixture of macromonomers or monomers and by changing the order of (co)polymerization in the formation of block-graft copolymers.

The spacing distribution is influenced by the relative reactivity ratios and the concentration of the macromonomer and the low molecular weight comonomer. The reactivity ratios are influenced by many complicated factors, but are mainly affected by:

- 1) the diffusion control effect, or the kinetic excluded volume effect, associated with the large size of a macromonomer;
- 2) the inherent reactivity of the terminal functionality on the macromonomer and the comonomer based on their chemical structure; and
- 3) the incompatibility effect potentially due to the thermodynamic repulsive interactions between a macromonomer and a propagating polymer chain.

The incompatibility effect is particularly important when preparing graft copolymers displaying micro-phase separation in the formed copolymer due to the chemical and structural differences between the macromonomer and the monomer.

The diffusion control effect was studied by Radke, W.; Müller, A. H. E. *Makromol. Chem., Macromol. Symp.* 1992, 54/55, 583. The copolymerization of a methyl methacrylate macromonomer with a methacrylate functional end group

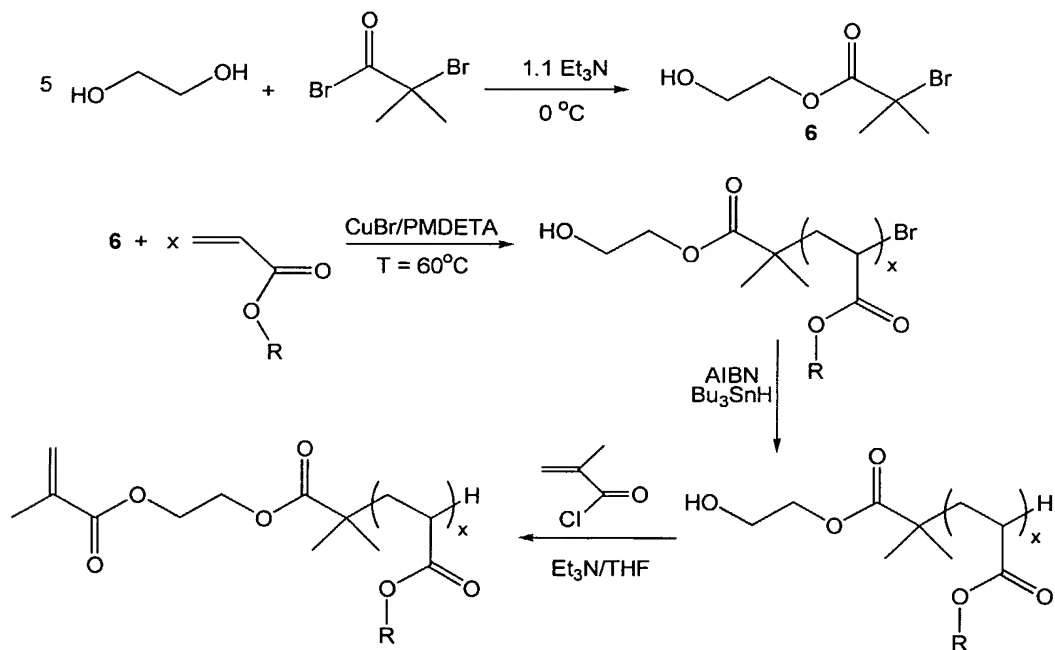
with methyl methacrylate was examined. In this case, due to the similarity in chemical structure of the macromonomer and monomer, factors 2) and 3) described above can be ignored. The reactivity ratio for methyl methacrylate, in such a copolymerization, using a conventional small molecule free radical initiator in a standard free radical polymerization process was determined to be between 1.12 and 2.14. They found that an increase in the macromonomer concentration would result in a reduction of the relative reactivity of the macromonomer. The increase in macromonomer increased the viscosity of the reaction medium and hence lowered the rate of diffusion of the functional end group on the macromonomer to the growing polymer chain end. This result indicates that the diffusion effect cannot be ignored in grafting through polymerization processes.

This phenomenon is exacerbated in the case of macromonomer copolymerization when there is a difference in compatibility between the macromonomer and backbone copolymer as described in factor 3. The incompatibility may be influenced by differences in polarity of the polymerization components, for example. In such a process, as the backbone polymer is formed the resulting solution of the backbone polymer and macromonomer in unreacted monomer, and optionally added solvent, may phase separate. In the phase separated polymerization medium, the reactivity of the macromonomer is significantly reduced and uncontrolled polymerizations may occur. Therefore, graft copolymers with heterogeneously distributed compositions along the copolymer chains are formed and the process does not produce well-defined copolymers. For certain specific applications, well-defined copolymers are desired.

The chemical composition of the branches and backbone of the graft (co)polymers are determined by the chemical composition of the initiator, the macromonomer and the monomer used in the process. Macromonomers and macroinitiators for use in the present invention may be prepared by any chemical process. Preferably, macromonomers may be prepared by controlled or living polymerization processes. Macromonomers may be prepared by controlled radical polymerization processes, such as ATRP utilizing a functional initiator.

Therefore, the micro- and macro-functionality may be incorporated into graft (co)polymers prepared by embodiments of the processes of the present invention.

A polymer or oligomer comprising a radically transferable atom or group prepared by ATRP, may be further used as a macroinitiator in an embodiment of the process of the present invention. Optionally, a polymer or oligomer comprising a polymerizable functional group and a radically transferable atom or group may be prepared by ATRP. The radically transferable atom or group may then be transformed into a non-reactive, non-transferable group before the polymer is further polymerized by a process of the present invention as a macromonomer. A general method for this process of preparing one such macromonomer is exemplified in Scheme 1. In the specific examples detailed in the experimental section a hydroxy functionalized initiator was used to polymerize *n*-butyl acrylate and *t*-butyl acrylate (polymerization of many other monomers have been described elsewhere). The halogen end group was then removed using tributyltin hydride and AIBN. The hydroxyl end group was then esterified with methacryloyl chloride to provide the polymerizable double bond.



Scheme 1

The reactions of Scheme 1 may be employed to provide at least three useful categories of functional materials applicable to the present invention. These

include a macroinitiator, a macromonomer, and a bi-functional macro-AB\*-monomer.

Prior to removal of the halogen, the first formed polymer comprises a hydroxy group and a radically transferable end group. This first formed polymer is an *active macroinitiator* that can be utilized for initiation of the copolymerization of macromonomers by ATRP or can be employed for further polymerization to form block copolymers which may be used as block co-macroinitiators. The block copolymers may then be converted into block-macromonomers by removal of the halogen, by the method of Scheme 1 or any other method, and esterification of the hydroxy-group remaining available on the initiator residue, using any known chemistry, such as, the procedures described below or the method described in Scheme 1.

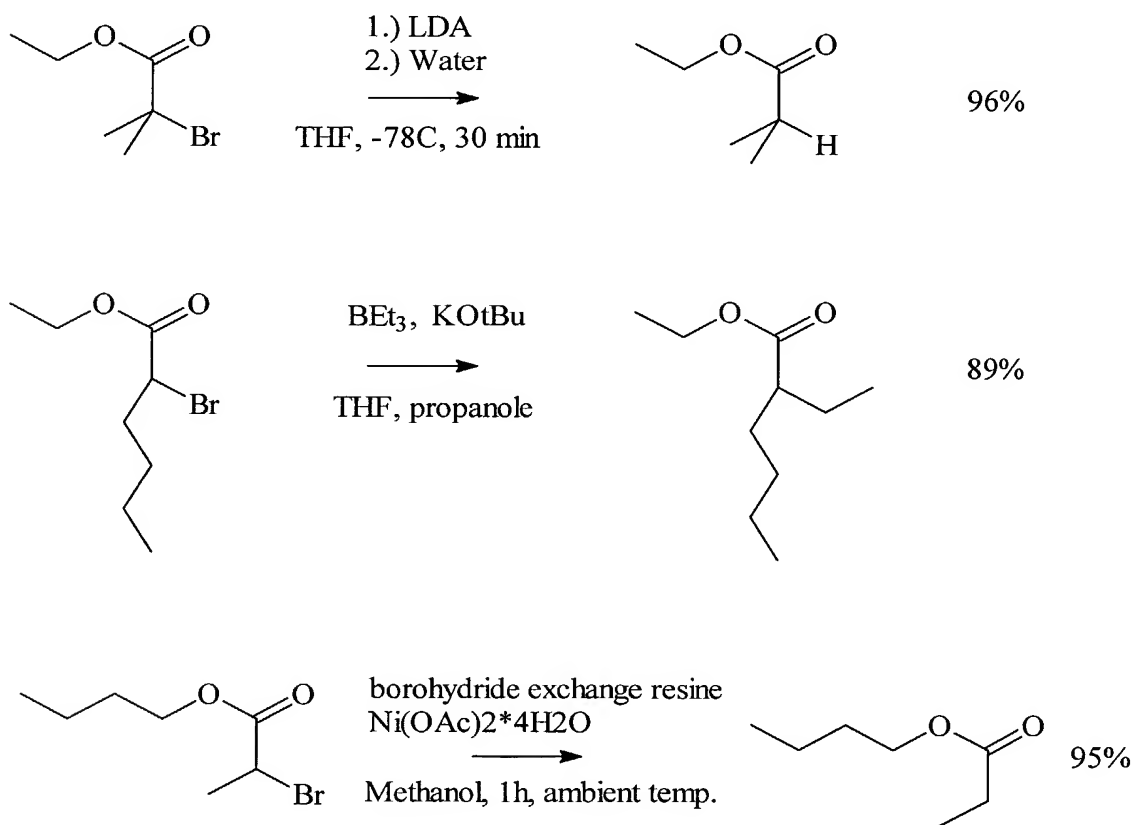
The resulting block-macromonomers may be incorporated in a polymerization process of the present invention initiated with a macroinitiator compatible with at least one segment of the block-macromonomer. The block macromonomers may be (co)polymerized by any polymerization process, such as, but not limited to, ATRP, ionic polymerization, standard free radical copolymerization or any other standard or controlled polymerization process incorporating a macroinitiator.

During the synthesis of macromonomers by ATRP for further polymerization by an embodiment of the process of the present invention comprising ATRP, the radically transferable end functionality on the first formed (co)polymer should, preferably, be substantially removed to avoid the formation of a bifunctional macromonomer, if this is not desired. More preferably, the radically transferable atom or group should be almost completely removed if one desires to form bottle brush polymers or fully controlled graft copolymers. Otherwise, the presence of such a bi-functional macromonomer may lead to the formation of (hyper)branched polymers during subsequent ATRP (co)polymerization. However, if desired, such materials prepared from a bi-functional macromonomer as a comonomer in a process of the present invention would be expected to produce useful materials, since the degree of branching in the final copolymer can be controlled by the molar ratio of functionality on the AB\* macromonomer. For



Example, such branched copolymers are known to have useful rheological properties.

In Scheme 1, a process for the conversion of the terminal bromo-atom into hydrogen using  $\text{Bu}_3\text{SnH}$  is shown. However, since tin hydride and the resulting tin bromide, may cause side reactions, such as termination or chain transfer, during the subsequent polymerization it should be at least substantially removed from the macromonomer prior to further polymerization. The macromonomer may be filtered through an absorbing media to remove the tin hydride and tin bromide. Other routes for bromine removal that avoid the problem of metal contamination or potential catalyst interactions are shown in Scheme 2.



Scheme 2

In the first reaction shown in Scheme 2, the desired product is obtained in high yield and the co-products, the lithium salt and diisopropylamine may be easily and

effectively removed from the macromonomer. This reaction may also performed to remove bromine from more sterically hindered  $\alpha$ -bromides.

Several inferences can be drawn from the embodiments disclosed herein comprising the ATRP of macromonomers that can be applied to other polymerization systems. These inferences apply to the application of the process of the present invention to other polymerization systems. First, while the rate of exchange and concentration of radicals remains the same in the controlled radical polymerization of macromonomers, the rate of polymerization is expected to be slow relative to copolymerization of small monomers using the same conditions.

In order to fully understand published conditions for ATRP and adapt them for copolymerization of macromonomers, embodiments of the process of the present invention were modeled using a simulation program, Predici. A conventional ATRP polymerization process was modeled including using one equivalent of a homogeneous copper-based catalyst complexed with a soluble bipyridine ligand, relative to a small molecule initiator, such as, ethyl 2-bromoisobutyrate. The kinetic data used to determine  $k_p$ , the polymerization rate constant; and  $k_t$ , the termination rate constant in the literature for Controlled Free Radical Polymerization (CFRP) was modeled. Subsequently, kinetic parameters were generated which identified conditions under which polymerization of a macromonomer could be controlled. The principles derived from this model were then extended to a model of an embodiment of the present invention comprising ATRP. The data generated from the Predici model was able to approximate the experimental data obtained for the copolymerization of a polystyrene macromonomer with a methacrylate end group with  $M_n = 4400$ . Using the initially proposed conditions the rate of propagation was slow. The rate of polymerization may be increase by using multiple equivalents of catalyst, as high as 10:1, to drive the polymerization forward. However, since  $k_d$ , the deactivation rate of the radical formed from the initiator, or growing polymer chain end, remains constant this would lead to a substantial decrease in initiation efficiency of a small molecule alkyl halide. We thought that this limitation could be avoided by using a macroinitiator such as those used to prepare block copolymers, thereby decreasing  $k_d$ , the deactivation rate constant, commensurately with  $k_t$ . Finally, to

limit the overall use of catalyst, a catalyst system with substantially higher  $k_a$  and/or lower  $k_d$  was implemented such that the radical concentration remains high under more 'catalytic' conditions.

With this fundamental understanding of the kinetic requirements for the incorporation of macromonomers into fully controlled ATRP polymerizations it was now possible to attest this insight and incorporate macromonomers prepared by any polymerization processes, preferably controlled polymerization processes, into the backbone of a free radically (co)polymerized polymer.

A series of radical copolymerization reactions were conducted with macromonomers and methyl methacrylate. This copolymerization process was chosen to allow direct comparison with the work of Radke and Mueller and exemplify without limiting the process of the present invention. In these embodiments, the backbone may comprise any free radically polymerizable (co)monomers and one only need consider the reactivity of the terminal functionality with the selected comonomers and composition of the macromonomer to attain similar results in different polymerization processes. Indeed such consideration is amplified in the discussion describing a process for the preparation of a graft (co)polymer with a homogeneous distribution of grafts. The preferred embodiment of the present invention comprises the atom transfer radical copolymerization of a compatible macroinitiator with a macromonomer. This approach led to enhanced incorporation of the macromonomer into the controllably, growing backbone polymer. In other radical polymerization processes where the rate of propagation for backbone copolymerization is faster, and less controlled than ATRP, there was a less pronounced, but still significant positive effect seen by use of a macroinitiator instead of a small molecule initiator. The other radical polymerization systems evaluated were RAFT, redox initiated polymerization, and conventional radical polymerization.

Embodiments of the process of the present invention described in the examples include the copolymerization of polydimethylsiloxane macromonomers with incompatible monomers forming an incompatible backbone to form graft copolymers to form graft copolymers. In the specific example provided of the (co)polymerization of polydimethylsiloxane, the macromonomer

and macroinitiator were prepared by a non-radical polymerization process. The copolymer selected to exemplify this approach comprised poly(methyl methacrylate) for the organic backbone polymer and poly(dimethylsiloxane) for the inorganic branches. Such organic/inorganic hybrid graft copolymers are of interest because polymers comprising inorganic substituents generally have unique and specific properties that organic analogues do not possess. Such hybrid graft copolymers exhibit good micro-phase separation and find use in a variety of applications, such as impact-resistant plastics, thermoplastic elastomers, compatibilizers and polymeric emulsifiers. Additionally, branched polymers comprising inorganic substituents generally show decreased melt viscosities which provides important advantages, especially for processing. Since the graft copolymers disclosed have many structural factors that can be varied (composition, backbone length, branch length, branch spacing, branch distribution, etc.), the process of the present invention has great potential to realize materials with new properties with high performance in specific applications.

A further example of the use of the process of the present invention is provided by the description of the preparation of a novel low viscosity graft copolymer with a backbone comprising a controlled molecular weight distribution and either heterogeneously or homogeneously distributed graft macromonomers prepared with biologically compatible and biologically degradable monomers. This example provides direction on the importance of the inherent reactivity of the end group and how end group reactivity can be manipulated in a controlled radical polymerization process to provide a graft copolymer with a narrow molecular weight distribution and selected distribution of grafts along a backbone chain. The graft copolymers prepared with biologically compatible grafts additionally exemplifies a process for development of bulk physical properties. The graft copolymer prepared by a process of the present invention comprises grafts that are soluble in the backbone copolymer at elevated temperature and additionally exemplifies a graft copolymer wherein the grafts comprise segments that can under certain circumstances crystallize thereby providing a phase separated bulk material while converting to a homogeneous material above the  $T_m$  of the

crystallizing phase. While the specific examples describe the preparation of a backbone polymer with a moderately high  $T_g$ , the procedure can be applied to copolymerization with monomers that provide polymers with low  $T_g$  matrices (i.e. below room temperature) and thereby can act as elastomers or impact resistant resins.

A further example of the level of macromolecular architectural control attainable from the process of the present invention is the preparation of block copolymers in which one or more of the blocks comprise a graft copolymer. The preparation of an AB and an ABC block-graft copolymers are described for specific examples but the approach can be applied to the preparation of block copolymers of any topology. The distribution of the graft segments in the backbone copolymer can be significantly modified by the order of formation of each graft-block copolymer segment. When such block copolymers, additionally comprising graft segments of differing composition, are to be prepared it is possible to select the composition of each graft, and the backbone so that the resulting copolymer can be homogeneous, or can phase separate in a bulk copolymer into two or three discrete phases. In the specific example provided in the experimental section the PDMS phase may phase separate due to the difference in phylicity between the graft and the backbone, and the PLLA, or a derivative of PLLA, can phase separate as a result of crystallization from the matrix, if the appropriate matrix composition is selected. The resulting triphasic morphology of a block-graft copolymer can be transitory, and dependent on the conditions employed for fabrication of the material, however once the desired morphology is obtained it can be stabilized by incorporation and reaction of additional reactive functionality on the grafts or backbone that allows crosslinking reactions to occur either within the copolymer or with added materials. Examples of such functionality have been described in earlier applications incorporated by reference but include glycidyl- functionality or silyl- functionality, the latter of which can be crosslinked by exposure to moisture. The crosslinkable functionality can be incorporated into one or more of the macromonomers or into the backbone during copolymerization or by transformation of a functional group on the copolymer after preparation of the material. The site of the reactive functionality

would impact the bulk properties of the formed resin while stabilizing the morphology.

Triphasic copolymers may also be prepared by the process of the present invention by preparation of copolymers with more than one  
5 macromonomer or by formation of gradient copolymers with more than one macromonomer.

Within the examples, specific conditions are provided for the copolymerization of macromonomers comprising an inorganic macromonomer, a biocompatible macromonomer, a polyolefin based macromonomer and the  
10 copolymerization of macromonomers of differing reactivity and macromonomers of differing composition. A biocompatible material is a material which is biologically compatible by not producing a toxic, injurious, or immunological response in living tissue or other undesirable effect. The types of macromonomers employed in the  
15 process described, since the improved incorporation of macromonomers by use of a compatible macroinitiator can be applied to any polymerization process and can encompass macromonomers comprising any polymerizable monomer unit.

The process of the present invention allows preparation of well-defined graft (co)polymers with either homogenously or heterogenously distributed  
20 branches. The distribution of the branches is primarily controlled by the relative reactivity ratio of the macromonomer and the monomer and the compatibilizing effect of the macroinitiator.

Copolymerizations of poly(dimethylsiloxane)-macromonomer and  
25 methyl methacrylate were performed in four different systems: a standard radical polymerization with a low molecular weight initiator and with a macroinitiator and an ATRP with a low molecular weight initiator and with a macroinitiator. To further highlight the beneficial effect of the use of a macroinitiator in other radical  
polymerizations processes, redox initiated polymerizations were also performed  
30 with a low molecular weight initiator and with a macroinitiator. The relative reactivity of the macromonomer in the copolymerization was compared for each polymerization.

In the example of a redox initiated polymerization, the copolymerization of methyl methacrylate and poly(dimethylsiloxane) macromonomer using a polydimethyl siloxane macroinitiator was conducted with a similar catalyst system to that employed for ATRP. The examples clearly indicate that a compatible macroinitiator increases the rate of incorporation of the macromonomer into the copolymer. The reactivity ratio of the macromonomer in the presence of a macroinitiator was 0.53. This can be compared to slower incorporation of the macromonomer, as measured by a reactivity ratio of 0.39 for the macromonomer, when no macroinitiator was employed. The reactivity ratio of this polymerization process is a little higher than that of the prior art conventional radical polymerization process. Polydispersity of the obtained copolymer was high ( $>2.7$ ). However, even in this example the beneficial effect of the macroinitiator is clearly seen in spite of the fact that the propagation rate is high and the effect of diffusion on the macromonomer can't be ignored.

The results of the experiments, detailed in the examples, indicate that in a controlled polymerization process the incorporation of the macromonomer into the polymer was greatly improved, viz. the higher reactivity ratio of macromonomer ( $1/r_1 = 0.85$ ) the controlled nature of the polymerization process significantly reduces, the effects of diffusion on the reactive macromonomer chain end.

Furthermore, use of a compatible macroinitiator can accelerate the incorporation of a macromonomer into the copolymer in the case of less controlled radical polymerizations in addition to fully controlled radical polymerization processes. The beneficial effect of the use of a compatible macroinitiator was most obvious when the polymerization was done at higher macromonomer concentration because the low molecular weight growing copolymer chain and macromonomer remain compatible.

As a further embodiment of the process of the present invention, conditions for incorporation of grafts comprising monomers from renewable natural sources into controlled polymerization processes have been disclosed. Such materials are of particular interest in the field of biologically interactive materials. In a non-limiting example, poly(methyl methacrylate)-g-poly(lactic acid)

graft copolymers were prepared by the process of the present invention.

Specifically, ATRP of methyl methacrylate and poly(lactic acid)-macromonomers are discussed. Graft copolymers with heterogeneous distribution of grafts with a gradient of grafted chains close to the chain end comprising the initiator residue

5 can be prepared or graft copolymers with heterogeneous distribution of grafts with a gradient of grafted chains close to the chain end comprising the growing polymer chain can be prepared or graft copolymers with a homogeneous distribution of grafts can be obtained when the functional end groups on the poly(lactic acid) macromonomers are selected to have an average reactivity ratio  
10 close 1.0. In the examples described below, for the preparation of a graft copolymer comprising a homogenous distribution of grafts the reactive end groups comprise an equimolar concentration of both methacryloyl and acryloyl functionality. Other functional end groups can be selected to allow similar "homogeneous" incorporation into backbone copolymers when different low  
15 molecular weight monomer units form the backbone copolymer or can be selected to afford the desired gradient heterogeneous distribution of grafts in a graft copolymer. The process of the present invention therefore comprises polymerizing a mixture of macromonomers, comprising different reactive end groups to control the rate of incorporation of the macromonomers into the graft  
20 (co)polymer.

The process for this control of incorporating macromonomers comprising different reactive end groups is exemplified by formation of a poly(lactic acid) graft copolymer. Poly(lactic acid) is a biodegradable aliphatic polyester derived from renewable resources and shows interesting properties:

25 Vert, M.; Schwarch, G.; Coudane, J. J. *Macromol. Sci., Pure Appl. Chem.* 1995, A32, 787-796; and Lipinsky, E. S.; Sinclair, R. G. *Chem. Eng. Prog.* 1986, 82, 26-32. It has been investigated as a biomaterial for controlled drug delivery, Jackanicz, T. M.; Nash, H. K.; Wise, D. L.; Gregory, J. B. *Contraception* 1973, 8, 227; and Makino, K.; Arakawa, M.; Kondo, T. *Chem. Pharm. Bull* 1985, 33, 1195;  
30 as a material for bioabsorbable sutures or fibers, Pennings, J. P.; Dijkstra, H.; Pennings, A. J. *Polymer* 1993, 34, 942; and as substrates for implants for bone fixation, Leenslag, J. W.; Pennings, A. J.; Boss, R. R. M.; Rozema, F. R.;

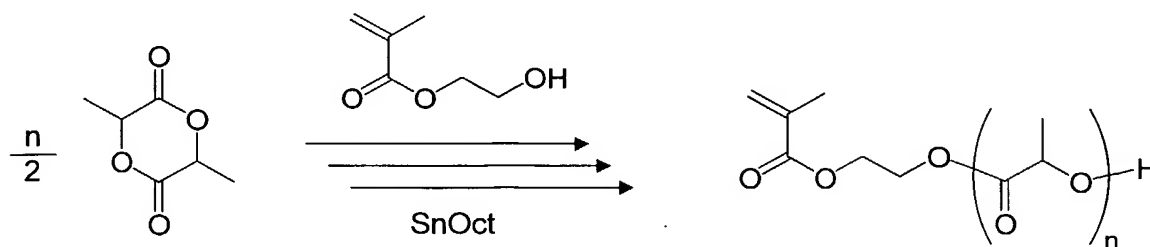


Boering, G. *Biomaterials* 1987, 8, 70; etc. Recently, as the production cost has been reduced because of technological breakthroughs, poly(lactic acid) is proving to be a viable alternative to petrochemical based plastics. Since graft copolymers are known to exhibit excellent properties, and have possibilities for a variety of applications, such as compatibilizers, emulsifiers, thermoplastic elastomers, or impact-resistant plastics, the study of precise structural control of graft copolymers containing poly(lactic acid) branches is of significant interest. One application for graft copolymers comprising such degradable graft segments would be in the preparation of porous membranes. The concentration of the grafted degradable polymer is selected so that the bulk graft-copolymer phase separated into a continuous bi-phasic morphology and then the degradable segment is removed allowing thereby forming a porous membrane.

Atom transfer radical copolymerization of methyl methacrylate and methacrylate-terminated poly(L-lactic acid) macromonomer (M-PLLA), methacrylate-terminated poly(D,L-lactic acid) macromonomer (M-PDLLA), and acrylate-terminated poly(L-lactic acid) macromonomer (A-PLLA) was investigated and is herein disclosed as one example of the present invention. The relative reactivity of the methacrylate-terminated macromonomer ( $1/r_{\text{MMA}} = 1.75$ ) and the acrylate-terminated macromonomer ( $1/r_{\text{MMA}} = 0.61$ ) were close to that of the small molecule models of the macromonomer end groups; 2-hydroxyethyl methacrylate and 2-hydroxyethyl acrylate, respectively and their individual copolymerization formed graft copolymers with a heterogeneous distribution of graft chains along the polymer backbone; one preferentially incorporated early in the polymerization and one incorporated later. This understanding allowed us to copolymerize a mixture of a methacrylate-terminated macromonomer and an acrylate-terminated macromonomer with methyl methacrylate by ATRP to give substantially homogeneously branched poly(methyl methacrylate)-g-poly(lactic acid) graft copolymer with low polydispersity ( $M_w/M_n = 1.15$ ) indicating a fully controlled copolymerization.

The poly(L-lactic acid) macromonomers containing either a methacryloyl end group, or an acryloyl end group, were prepared by the ring opening polymerization of L-lactide using a tin catalyst, and using 2-hydroxyethyl

methacrylate or 2-hydroxymethyl acrylate as the initiator, Scheme 3 shows the use of 2-hydroxyethyl methacrylate.



Scheme 3

5

Highly functionalized, low polydispersity poly(L-lactic acid) macromonomers were obtained in both cases. Although the molecular weight from GPC was higher than the value from NMR, it is consistent with our experience ( $M_n$  from GPC is about 1.5 times higher than actual molecular weight). The macromonomers were obtained as white powders, soluble in chloroform, THF, hot toluene, or hot xylene.

10

The ATRP copolymerization of poly(L-lactic acid) macromonomer with a methacrylate end group and methyl methacrylate was performed at 90°C. The composition of poly(L-lactic acid) macromonomer in feed was 3.5 mol %, or 50 wt %. The target degree of polymerization was 300. A homogeneous polymerization was conducted in the presence of a mixed diphenyl ether/xylene solvent and the poly(L-lactic acid) macromonomer was efficiently incorporated into the copolymer. The resulting reactivity ratio of methyl methacrylate ( $r_{MMA}$ ) was calculated to be 0.58. This indicates a gradient copolymer was formed with preferential incorporation of the grafts closer to the terminus of the backbone containing the attached initiator residue.

15

20

Use of an acrylate functionality on the poly(L-lactic acid) macromonomer gave a copolymer in which the resulting reactivity ratio of methyl methacrylate ( $r_{MMA}$ ) was calculated to be 1.63. This again indicates the copolymer has a heterogeneous distribution of grafts or branches and that a gradient copolymer was formed, in this case with preferential incorporation of the grafts closer to the terminus of the backbone containing the radically transferable atom; i.e. the opposite of the first gradient copolymer prepared with the methacrylate

25

end group. This led to the understanding that copolymerization of both macromonomers with methyl methacrylate can provide for the preparation of a graft copolymer with homogeneous distribution of grafts along the backbone chain and provide for a backbone chain comprising a narrow molecular weight distribution.

Furthermore, use of the appropriate functionality on the poly(L-lactic acid) macromonomer can allow preparation of an elastomeric material; one with an acrylate backbone, that comprises the matrix, and a crystallizable graft, that is further susceptible to biological degradation during recycle processes. In the specific case described in the examples the acrylate selected for the exemplary backbone copolymer is butyl acrylate. This interesting graft copolymer has a soft segment backbone and biodegradable hard segments, which have the ability to undergo phase separation and aggregation through crystallization. Such a graft copolymer may be useful as a biodegradable thermoplastic elastomeric material.

The graft copolymers prepared above are examples of the type of materials that can be prepared by an embodiment of the present invention by a controlled radical polymerization process such as, but not limited to, ATRP, NMP or RAFT copolymerization, of a macromonomer with radically (co)polymerizable monomer(s).

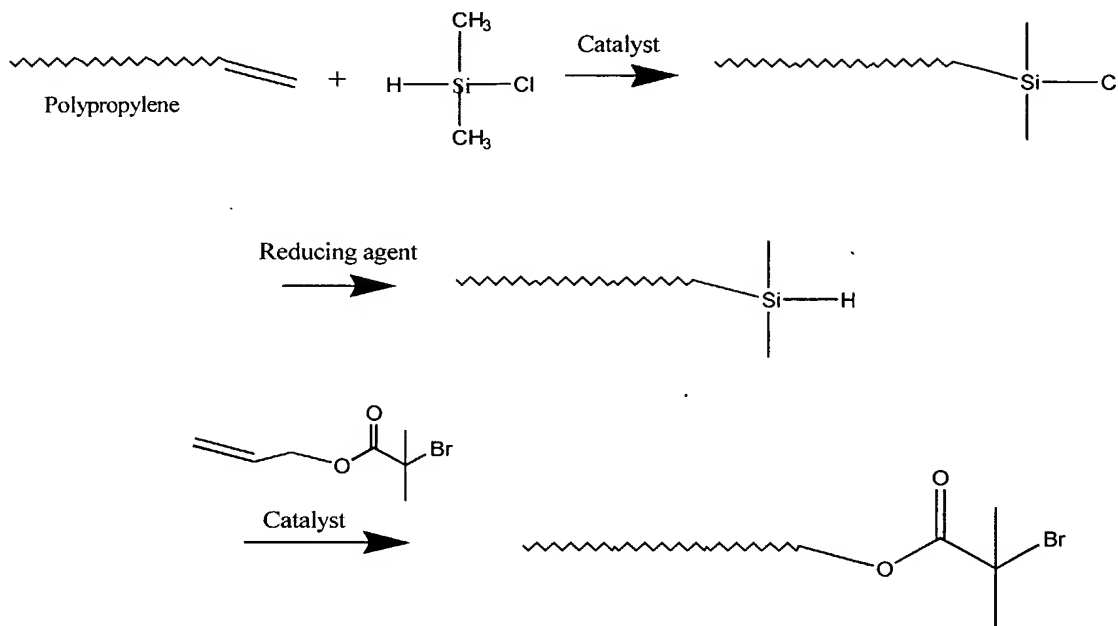
A further extension of this control over the composition of graft copolymers is the copolymerization of at least two macromonomers with different compositions with one or more low molecular weight monomers. The result of such a copolymerization would be a graft copolymer with grafted polymer chains comprising two different compositions attached to the same backbone (co)polymer. In the examples below, gradient-graft-copolymers were prepared by the process of the present invention wherein the distribution of each macromonomer along the backbone copolymer was heterogeneous, however, by applying the principles discussed above the graft distribution of each macromonomer can be changed. The composition of the components of such a material can be selected to allow phase separation into a constrained bi-phasic or tri-phasic structure.

In a further series of embodiments of the present invention graft copolymers were used as macroinitiators for the (co)polymerization of macromonomers. Block-graft-copolymers were prepared wherein each block comprises a graft copolymer. This approach to material synthesis can also lead to constrained bi-phasic or tri-phasic structures in the resulting bulk copolymer. This series of experiments were conducted first to further exemplify the range of materials capable of being prepared by this technology, and further to provide materials for the examination of properties of such potential tri-phasic systems. In the examples below the preparation of an AB and an essential ABC block-graft copolymer are described. The physical properties or the morphologies of such tri-phasic graft copolymeric materials have not yet been explored but one would expect, if appropriately selected, for them to act in concert. In one exemplary case, one phase may act as a toughening agent while another provided channels of selective permeability in an article or film. A potential application for the exemplary material prepared below in the examples would be for one graft, for example, a PLLA macromonomer based graft, to provide biocompatibility, or selective permeability and for another graft, for example a PDMS graft, to provide increased toughness. Appropriate selection of the graft segments could also provide for a toughened matrix material with cylindrical or gyroidal channels for proton transfer.

A further series of examples builds on this recognition that the end group reactivity has to be considered when constructing graft copolymers. The copolymerization of (meth)acrylate terminated polyethylene macromonomers and (meth)acrylates and (meth)acrylate terminated polypropylene macromonomers providing graft copolymers with polyolefin grafts is described. Such graft copolymers can act as surfactants in stabilizing blends of polymers found in municipal recycle streams providing for improved "polymeric wood" formed by melt blending such streams.

Two approaches were evaluated for the preparation of polypropylene macroinitiators starting with polypropylene macromonomers comprising a terminal vinyl group. In the first route the polypropylene was

functionalized using chlorodimethylsilane. This first route to the polypropylene-based macroinitiator is presented in Scheme 4.

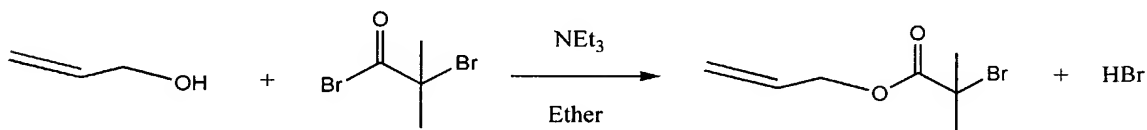


5

Scheme 4: First route for the synthesis of polypropylene-based macroinitiator.

Within this reaction scheme there is an extra step, the synthesis of allyl-2-bromoisobutyrate. The reaction scheme for this step is presented in Scheme 5.

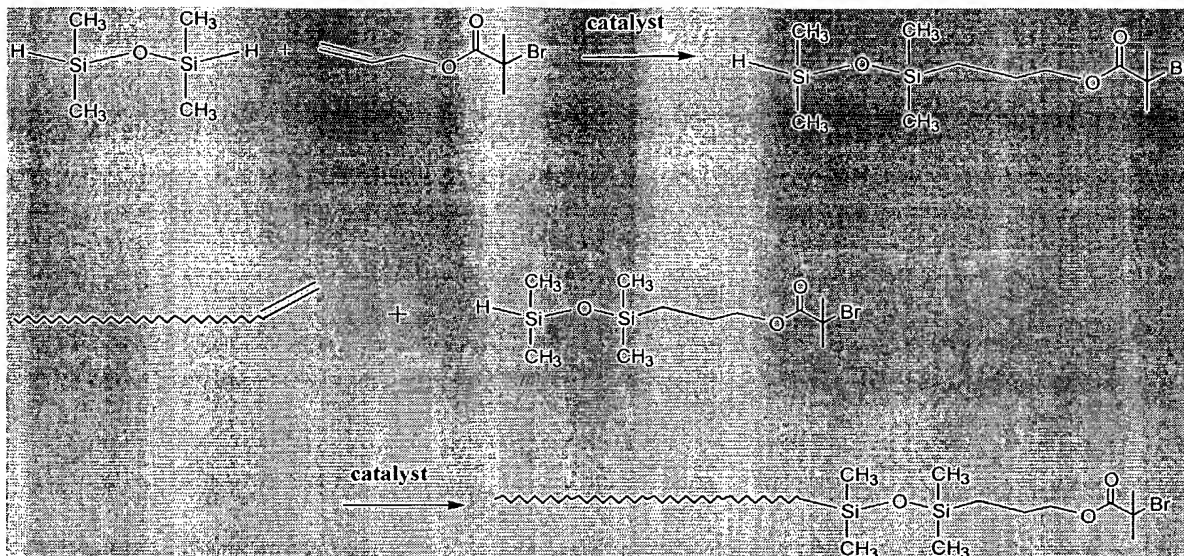
10



Scheme 5: Synthesis of allyl-2-bromoisobutyrate.

A second route started with the functional initiator prepared by the first reaction in Scheme 4 and used tetramethyldisiloxane to form a linking species. The second route is presented in Scheme 6.

15

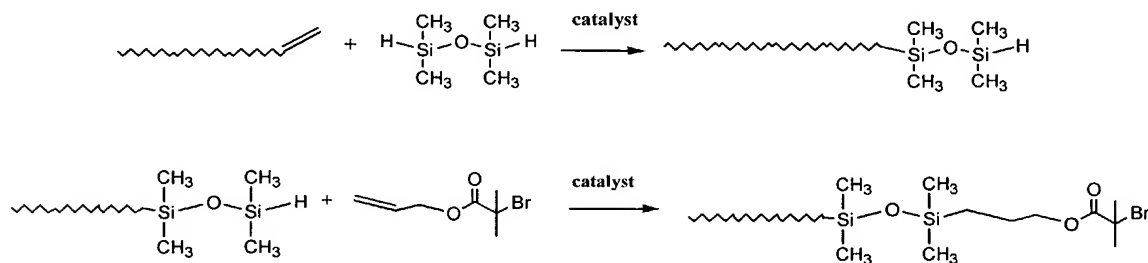


Scheme 6: Second route for the synthesis of polypropylene-based macroinitiator.

This route permits one to avoid the reduction step. By avoiding the  
 5 reduction step, the coupling reaction with the unsaturated polypropylene is the  
 final step, thus reducing the number of steps involving the latter species from  
 three to one. The major drawback of the reaction in Scheme 6 is that it leads to a  
 polymer with a siloxane bond, which is not the case of the route described in  
 Schemes 4 and 5 using chlorodimethylsilane. Consequently, copolymers  
 10 prepared from the macroinitiator prepared according to Scheme 6 have a lower  
 stability toward strong acids and bases.

A variation on this second route for the preparation of the  
 polypropylene macroinitiator was also evaluated. Platinum complexes can  
 catalyze the formation of a Si-Si bond during hydrosilation reactions; Brown-  
 15 Wesley, K. "Formation of Si-Si Bonds in the Presence of Hydrosilation Catalysts"  
*Organometallics* 1987, 6, 1590. This formation is due to a possible bimolecular  
 reaction between two silane entities, leading to a disilicon compound and to the  
 formation of H<sub>2</sub>. This proportion of products formed by such a side reaction may  
 increase with the concentration of silane in the reaction medium, as well as with  
 20 the temperature of the reaction; Britcherm L.G., Kehoe, D.C.; Matisons, J.G.,  
 Swinser, A.G. "Siloxane Coupling Agents" *Macromolecules* 1995, 28, 3110-3118.  
 In the procedure discussed above the reaction between tetramethyldisiloxane and

allyl-2-bromoisobutyrate is conducted in bulk, with a huge excess of silane (15 : 1) and at a temperature of 50 °C. The first hydrosilation between vinyl-terminated polypropylene and tetramethyldisiloxane allows us use of a smaller excess of silane, as the concentration of unsaturated chain end is very low. Moreover, the reaction can be carried out in xylene and at a temperature close to room temperature. The reaction scheme, with inverted order of addition, is presented in Scheme 7.



Scheme 7: Variant of the second route with inverted order.

Both approaches allow the preparation of suitable macroinitiators to allow the successful preparation of block copolymers with an A block comprising polypropylene or a graft copolymer comprising improved incorporation of polypropylene macromonomers, according to the process of the present invention.

The use of a compatible macroinitiator to control the efficiency of initiation of a copolymerization also operates in inherently biphasic systems. When a catalyst for ATRP is dissolved in an ionic liquid and dispersed in a polymerization mixture containing a low molecular weight initiator the initiator initially migrates to the ionic liquid, forms a high concentration of radical species which undergo termination reactions, thereby leading to low initiator efficiency. However if a macroinitiator is employed the initiator does not migrate to the ionic liquid phase and high initiator efficiency is seen when the catalyst diffuses to the polymerization phase. This observation allows the use of environmentally benign ionic liquids in place of organic solvents in the preparation of graft copolymers and thereby provides a polymerization process that has lower environmental impact.

A "living" polymerization is a polymerization process where termination reactions are minimal. As used herein, "control" and/or "controlled" in reference to

polymerization processes means that the polymerization process conditions are defined whereby the contributions of the chain breaking processes are insignificant compared to chain propagation, so that polymers with predetermined molecular weights, low polydispersities and tailorable, selected functionalities are achievable. The features of a controlled polymerization are well known and have been described in detail in application 09/972,260, which is herein incorporated by reference. For the present purposes "polymers" include homopolymers and copolymers (unless the specific context indicates otherwise), which may be block, random, statistical periodic, gradient star, graft, comb, (hyper)branched or dendritic. The "(co)" parenthetical prefix in conventional terminology is an alternative, viz., "(co)polymer" means a copolymer or polymer including homopolymer while "(co)polymerizable" means a monomer that is directly polymerized by the polymerization mechanism being discussed and additionally a comonomer which can only be incorporated into the polymer by copolymerization. Similarly "(hyper)" is meant to incorporate the concept that the degree of branching along the polymer backbone can vary from a low degree of branching up to a very high degree of branching.

## EXAMPLES

### *Analytical Measurements*

GC: Monomer conversion was determined using a Shimadzu GC 17A gas chromatograph equipped with an AO20ic autosampler and a FID detector using J&W Scientific 30 m DB608 column. Injector and detector temperatures were kept constant at 250°C. (Conditions: for butanone (internal standard) / n-butyl acrylate (monomer): start temperature 50°C, isotherm 0 min., heating rate 30°C/min., final temperature 160°C, isotherm 0 min.; for toluene (internal standard) / styrene (monomer): start temperature 40°C, isotherm 1 min., heating rate 20°C/min., final temperature 130°C, isotherm 0 min.) The conversion was calculated by detecting the decrease of the monomer peak area relative to the standard peak area. GC measurements were repeated 3-7 times for each sample to reduce errors.



*GPC:* Polymer molecular weights were estimated by a GPC equipped with a Waters WISP 712 autosampler, column setup (guard,  $10^2$  Å,  $10^3$  Å,  $10^5$  Å; 5 µm; Polymer Standard Service (PSS), Germany), and a Waters 410 RI detector in THF (35°C), at a flow rate of 1 mL/min. The setup was calibrated against low polydispersity poly(methyl methacrylate) (pMMA) or polystyrene (pS) standards (PSS, Germany) with toluene as internal standard. Weight average molecular weights were determined by using size exclusion chromatography equipped with Waters microstyragel columns (pore size  $10^5$ ,  $10^4$ ,  $10^3$  Å) in THF as a solvent) a differential refractometer (Waters Model 410), light-scattering detector (DAWN Model F), a differential viscometer (Viscotek Model H502) and GPC Win software.

*NMR:*  $^1\text{H}$ -NMR characterization was performed on a Bruker 300 MHz spectrometer. The measurements were carried out in chloroform- $d_3$  or methylene chloride- $d_2$ . The amount of incorporated monomer in nBuA/S copolymers was determined by  $^1\text{H}$ -NMR measurements by comparison of the peak area ratio of characteristic signals for pnBuA (0.93 ppm, t,  $^3J = 7.3$  Hz, 3H, O-(CH $_2$ ) $_3$ -CH $_3$  and ppm = 4.00, bs, 2H, O-CH $_2$ -CH $_2$ -CH $_3$ ) and pS (7.23 - 6.10 ppm, m, 5H, Ar-H).

*AFM:* Scanning force micrographs were recorded at ambient conditions with a Nanoscope III instrument (Digital Instruments, St. Barbara, USA) operating in the tapping mode. The measurements were performed at ambient conditions (in air, 56% RH, 23°C) using Si cantilevers with a spring constant of  $\approx$  50 N/m, a tip radius of 8 nm, and a resonance frequency of about 300 kHz. The set-point amplitude ratio was maintained at 0.9 to minimize the sample deformation induced by the tip. The samples for tapping mode SFM measurements were prepared by spin casting at 2000 rpm of dilute solutions (0.15 g/l) of brush molecules in chloroform.

*Evaluation of molecular weights:* Degrees of polymerization were estimated from conversion by assuming quantitative initiation for both initiator and macroinitiator,  $\text{DP}_{\text{conv}} = \Delta[\text{M}]/[\text{I}]_0$ . Molecular weights were estimated using GPC with refractive index detector (linear polystyrene or poly(methyl methacrylate)

standards as well as using MALLS detector and universal calibration with LS and viscometry detectors.

1. *General procedure for macroinitiator and macromonomer preparation.*

The general procedure for preparation of macroinitiators and macromonomers was to polymerize *n*-butyl acrylate and *t*-butyl acrylate with a hydroxyl functionalized initiator. These first formed polymers may be macroinitiators for ATRP before the halogen end group is removed. Optionally, the halogen end group may be transformed into a functionality capable of initiating any polymerization process. For production of a radically polymerizable macromonomer, the halogen end group may be removed using tributyltin hydride and AIBN. The hydroxyl end group may then be esterified with methacryloyl chloride to provide a polymerizable double bond.

A series of macroinitiators and macromonomers were prepared using this approach and the molecular weights are listed in Table 1.

Table 1: Molecular weight data for polymers for use as macromonomers and macroinitiators.

Sample	$M_n^*$	$M_w/M_n$	$DP_n^{**}$
pnBA macromonomer	3300	1.12	22
ptBA macromonomer	2900	1.13	18
ptBA macromonomer	5900	1.29	45

\*based on GPC for final coupled/dehalogenated product.

\*\*based on GPC data for initial (HO-pBA-Br) sample. Little or no variation was observed as polymer was transformed.

End group analysis was carried out on the first formed polymers, the macroinitiator form, using  $^1\text{H}$  NMR and ESI-MS. In the macroinitiator form, the resonances from the methine proton adjacent to the halogen end group (4.1 ppm) and the terminal hydroxyl group (3.1 ppm) are not detectable. However, after conversion to a macromonomer the vinyl protons are clearly visible (5.6 and 6.2 ppm).

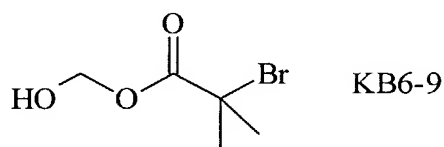
ESI-MS also revealed the presence of a tin dimer ( $M/Z = 600 \pm 1$ ). The molecular weight of the macromonomer that was injected exceeded the detection limit of the instrument, however, the low molecular weight edge of the

distribution is visible and shows the distribution of end group functionalities clearly. There is no evidence of residual halogen terminated polymer ( $m = 132 + 128.2n + 79/81 + 23$ ) after reaction with tin bromide. The main series of peaks correspond to the sodium salt of the polymer with a saturated end group ( $m = 132 + 128.2n + 1 + 23$ ) along with sister peaks corresponding to loss of one and two hydrogen atoms, respectively. After the esterification reaction with methacryloyl chloride, the main series of peaks correspond to the macromonomer product ( $m = 200 + 128.2 + 1 + 23$ ) in addition to the same sister peaks with one and two fewer hydrogen atom(s). While there is considerable noise, there is no evidence of residual unfunctionalized polymer.

The first polymers prepared above may also be used as macroinitiators prior to removal of the radically active halogen on the chain end. They can therefore also be used for the preparation of block macromonomers by chain extension with different free radically polymerizable monomers prior to removal of the first halogen and esterification of the hydroxy group on the other polymer terminus.

1A. *Macroinitiator, or Macromonomer precursor, (HO-ptBA-Br):*

The HO-functionalized initiator shown below, KB6-9; was used to synthesize a mono-hydroxyfunctionalised poly-t-butyl acrylate (*HO-ptBA-Br*). The reaction conditions had been defined by the polymerization of t-butyl acrylate using  $\alpha$ -bromo-isobutyrate ethylester as initiator.



Experimental conditions:

ratio of reagents; [KB6-9]:[t-BA]:[CuBr]:[CuBr<sub>2</sub>]:[PMDETA] = [1]:[67]:[0.25]:[0.0125]:[0.25]; solvent, acetone (20 vol%); reaction temperature 60°C; conversion: 55% (GC) after 4 h.

Results:  $M_{n,th} = 4700$ ;  $DP_{n,th} = 37$ .  $M_n = 4500$ ;  $DP_n = 35$ ; MWD = 1.17.

1B. *Macroinitiator, or Macromonomer precursor (HO-pMMA-Br)*

Experimental conditions:

Ratio of reagents; [KB6-9]:[MMA]:[CuCl]:[dNbpy] = [1]:[250]:[0.75]:[1.5];

Anisole 2 vol%; reaction temperature 90°C. Conversion, 83% (GC) after 9

h. Results:  $M_{n,th}$  = 25000;

5 *1C. Block Co-Macroinitiator, or Co-Macromonomer precursor (HO-ptBA-block-pSt-Br)*

A block co-macroinitiator is a macroinitiator comprising a block copolymer and a block co-macromonomer is a macromonomer comprising a block copolymer. The preparation of a block co-macroinitiator or co-macromonomer precursor was accomplished by chain extension using the HO-ptBA-Br prepared in example 1A, as a macroinitiator to further polymerize styrene thereby preparing a block copolymer macroinitiator, or a block-co-macromonomer precursor.

Experimental conditions:

ratio of reagents; [HO-(ptBA)<sub>35</sub>-Br]:[styrene]:[CuBr]:[PMDETA] =

15 [1]:[100]:[1]:[1];

reaction temperature 100°C.

Conversion: 96% (GC) after 140 min.

The GPC additionally displayed a clean shift to higher molecular weight.

Results:  $M_{n,th}$  = 15000;  $DP_{n,th}$  = 37+100.  $M_n$  = 18300; MWD = 1.11.

20 *1D. Substitution reaction (Br  $\leftrightarrow$  H) Conversion of a macroinitiator to a macromonomer.*

A possible route to change the bromo atom into hydrogen is to use Bu<sub>3</sub>SnH. The tin hydride and the products of the reaction, (tin bromide) may cause side reactions (termination or chain transfer) during the ATRP polymerization. To avoid this problem of metal contamination, or potential catalyst interactions various other reactions, shown above in Scheme 2, can be used for the transformation:

25 *2. PMMA-g-poly(dimethylsiloxane) Graft Copolymers*

PMMA-g- poly(dimethylsiloxane) (PDMS) graft copolymers samples were prepared using ATRP, RAFT, redox initiated polymerization and a conventional radical polymerization.

30

*Materials.* Copper chloride (CuCl) (98%, Aldrich) was purified by known procedures; 4,4'-Di-n-nonyl-2,2'-bipyridine (dnNbpy) was synthesized by a modified disclosed literature procedure. Methacrylate-terminated PDMS macromonomer (PDMS-MA) was prepared by the ring opening  
5 polymerization of hexamethylcyclotrisiloxane (D3) (98%, Aldrich) followed by reaction with 3-methacryloxypropyldimethylchlorosilane.  $M_n = 2370$  (toluene-GPC),  $M_w/M_n = 1.25$ , the functionality of the terminal methacrylate group was 1.0 (H-NMR, in CDCl<sub>3</sub>).

*Measurements.* The conversion of MMA was measured using a Shimadzu  
10 GC-14A gas chromatograph (GC) equipped with a widebore capillary column (30 m, DB-Wax, J&W Sci.). The conversion of PDMS-MA was determined by the gel permeation chromatography measurements in toluene (toluene-GPC) using a Waters 510 liquid chromatograph pump (1 ml/min, 30°C) equipped with columns (guard, 10 5 Å, 10 2 Å; Polymer  
15 Standards Service) in series with a Waters 2410 differential refractometer. [5] The molecular weight of the polymer was measured by GPC in tetrahydrofuran (THF-GPC) based on linear poly(methyl methacrylate) standards using a Waters 515 liquid chromatograph pump (1 ml/min, 30°C) equipped with four columns (guard, 10 5 Å, 10 3 Å, 100 Å; Polymer  
20 Standards Service) in series with a Waters 2410 differential refractometer.

*2A. Preparation of PDMS-Macromonomer with Terminal Methacrylate Functionality.*

In a dry box, D3 (24.5 g, 0.11 mol) was placed into a 100 mL round bottom flask and dissolved in a mixture of THF (8 mL) and toluene (8 mL). n-  
25 Butyllithium in hexane (4 mL, 10 mmol) was added to this clear, colorless solution and the resulting solution was stirred for 2 hours at room temperature. 3-Methacryloyloxypropyldimethylchlorosilane (2.65 g, 12 mmol) predissolved in hexanes (1.5 mL) was added to the solution and the heterogeneous white mixture was stirred at room temperature overnight. All the volatile materials were  
30 removed at room temperature under vacuum (1 mmHg) over 5 hours and the mixture was stored in the freezer overnight. The mixture was diluted in 20 mL of hexanes and filtered to remove all salts. The solvent was removed from the

filtrate by rotary evaporation and any remaining 3-methacryloyloxypropyldimethylchlorosilane was removed under  $1 \times 10^{-7}$  mmHg vacuum at 60 °C for four hours. The clear liquid was then stored in the freezer. Yield = 17.6 g (72%).  $^1\text{H-NMR}$  ( $\text{CDCl}_3$ )  $\delta$  : 6.1 (s), 5.5 (s), 4.1 (t), 1.9 (s), 1.7 (m), 1.3 (m), 0.9 (t), 0.6 (m), 0.1 (s). Methacrylate terminal group functionality (F) was calculated from the relative peak intensity in the  $^1\text{H-NMR}$  spectrum at 1.9 ppm (methyl 3H in methacrylate group) and 0.9 ppm (methyl 3H in butyl group) and was 1.0.  $M_n = 2200$ ,  $\text{PDI} = M_w/M_n = 1.18$  (determined by GPC in THF, PMMA calibration).

#### 2B. Preparation of PDMS-Macroinitiator.

A PDMS-macroinitiator containing a 2-bromoisobutyrate terminal group was synthesized by anionic polymerization of D3 terminated by chlorodimethylsilane, followed by reaction with 3-butenyl 2-bromoisobutyrate. Yield, 78%.  $^1\text{H-NMR}$  ( $\text{CDCl}_3$ )  $\delta$  : 4.2 (t), 1.9 (s), 1.7 (m), 1.5 (m), 1.3 (m), 0.9 (t), 0.6 (m), 0.1 (s). The functionality ( $F = 0.95$ ) was calculated by comparison of the integrations of the methyl protons in bromoisobutyrate group at 1.9 ppm with the methyl protons in n-butyl group at 0.9 ppm in the  $^1\text{H-NMR}$  spectrum.  $M_n = 15,600$ ,  $M_w/M_n = 1.10$  (determined by GPC in THF, PMMA calibration).

#### 3. ATRP Copolymerization Initiated by Ethyl 2-Bromoisobutyrate.

A typical polymerization is as follows: a PDMS-macromonomer ( $M_n = 2200$ ,  $\text{PDI} = 1.18$ ,  $F = 1.0$ , 0.95 g, 0.43 mmol) was added to a 25 mL Schlenk flask equipped with a stir bar and was deoxygenated by degassing overnight and back-filling with nitrogen.  $\text{CuCl}$  (2.9 mg, 0.029 mmol) and  $\text{dnNbpy}$  (23.5 mg, 0.058 mmol) were placed in a 25 mL round bottom flask equipped with a stir bar. This flask was capped with a rubber septum, purged with nitrogen gas for half an hour, then charged with (0.82 g, 8.2 mmol) and deoxygenated xylene (0.95 mL) via syringes. The mixture was stirred at room temperature under nitrogen for about half an hour until a homogeneous maroon solution formed. This solution was cannula-transferred to the Schlenk flask under a nitrogen flow. Ethyl 2-bromoisobutyrate (EBiB) (4.2  $\mu\text{L}$ , 0.029 mmol) was added to the flask through the capped arm on the flask. A 0.1 mL aliquot of the reaction mixture was removed and the flask was placed into a 90 °C oil bath under nitrogen. Periodically, 0.1 mL

aliquots of the reaction mixture were removed for kinetic and molecular weight analysis. These samples were diluted with toluene to 30 wt%, subjected to GC to measure the MMA conversion, toluene-GPC to measure the macromonomer conversion and THF-GPC to determine the molecular weights of the generated copolymer. The results of ATRP copolymerization of MMA and a polydimethylsiloxane-macromonomer (PDMS- macromonomer) at 90 C are shown in Table 2.

Table 2. ATRP copolymerization of MMA and PDMS-Macromonomer

Exp. No.	MMA, g	p-Xylene, wt ratio to MMA	PDMS-MM, g	M/MMI/CuCl/Ligand mol ratio	Temp., C	Time, hr	Yield, g	$M_n$	$M_w/M_n$
HS166	2.24	1 : 1.2	2.8	285/15/1/1/2	90	21.6	3.3	86,020	1.58

Macromonomer: HS134,  $M_n$ =2370

Initiator: ethyl 2-bromoisobutyrate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

Copolymer was isolated by reprecipitation (MeOH/Acetone).

$M_n$  and  $M_w/M_n$  were calculated from THF-GPC.

GPC traces showed slow consistent incorporation of the macromonomer into the copolymer. The kinetics of the copolymerization and the Jaacks plot for the copolymerization are shown in Figure 2. The  $r_{MMA}$  was 1.34.

After the reaction was completed, the reaction mixture was diluted with acetone, treated with ion change resin and passed through an alumina column. The solution was poured into methanol to precipitate the copolymer. Dissolution and reprecipitation from acetone/MeOH was repeated twice. About 3g of copolymer were recovered. Toluene-GPC indicated that most of the unreacted PDMS-macromonomer was removed by the precipitation process. Copolymer contained less than 2 wt% of unreacted macromonomer.

### 3A. ATRP Copolymerization Initiated by PDMS-Macroinitiator.

A typical polymerization is as follows: A 25 mL Schlenk flask equipped with a stir bar containing a PDMS-macromonomer ( $M_n$  = 2200, PDI = 1.18, F = 1.0, 0.84 g, 0.38 mmol) and a PDMS-macroinitiator ( $M_n$  = 15600, PDI = 1.10, F = 0.95, 0.42 g, 0.025 mmol) was evacuated overnight and back-filled with nitrogen. In a 25 mL round bottom flask, a stir bar, CuCl (2.5 mg, 0.025 mmol),

and dnNbpy (20.7 mg, 0.051 mmol) were placed. This flask was capped with a rubber septum, purged with nitrogen gas for half an hour, then deoxygenated MMA (0.72 g, 7.2 mmol) and deoxygenated xylene (1.62 mL) were added via syringes. The mixture was stirred at room temperature under nitrogen for about half an hour until a homogeneous maroon solution formed. This solution was cannula-transferred to the Schlenk flask a under nitrogen flow. The polymerization was performed at 90 °C and periodic sampling and measurements were done in the same way as in the case of the EBiB initiated reaction described above.

10 *3B. ATRP Copolymerization of MMA and PDMS-Macromonomer (75°C, in solution)*

Since almost all of the ATRP copolymerizations had been run at 90 °C, in order to compare directly with the RAFT system, an ATRP polymerization was run at 75 °C.

Exp. No.	MMA, g	p-Xylene, wt ratio to MMA	Macromonomer (SiM) <sub>2</sub> , g	M/SiM/I/CuCl/Ligand mol ratio	Temp., C
HS160	0.77	1 : 1	0.96	285/15/1/1/2	75

Macromonomer: HS134, Mn=2370(DP32), PDI=1.25, F=1.0

Initiator: ethyl 2-bromoisobutyrate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

15

The reactivity ratio  $r_{\text{MMA}}$  was found to be 1.37. This value was slightly higher than ATRP at 90 °C. This can be attributed to the increased solution viscosity at lower temperature.

*4. RAFT Polymerization of MMA*

20

Materials. 2-Phenylprop-2-yl dithiobenzoate (cumyl dithiobenzoate, (CDB)) was synthesized by the reaction of dithiobenzoic acid with  $\alpha$  - methylstyrene in carbon tetrachloride at 70 °C for 4 hours and then purified by column chromatography on alumina with hexane as eluent. The dithiobenzoic acid was prepared by the reaction of benzyl chloride with elemental sulfur and



sodium methoxide. Benzoyl peroxide (BPO) (Fisher, 75%) and 2,2'-azobisisobutyronitrile (AIBN) (98%, Aldrich) were used as received.

Polymerizations. The conditions for the RAFT copolymerization of MMA and PDMS macromonomer were defined by conducting a series of scoping experiments to determine effective reaction conditions. In the initial experiment MMA was polymerized by RAFT method at 75 °C with AIBN as initiator and 2-phenylprop-2-yl dithiobenzoate (cumyl dithiobenzoate, (CDB)) as addition-transfer fragmentation agent in their molar ratio of MMA/AIBN/CDB=300/1/0.5. (Run HS156.) The molecular weight of the resulting copolymer was higher than the theoretical line. This was attributed to the rather high reaction temperature leading to rapid decomposition of the AIBN so that many radicals were initially generated and transfer agent (CDB) didn't work well.

Therefore, conditions similar to those reported by Rizzardo in the literature were used, except the reaction was run in xylene not in benzene (HS157-1). Two additional scoping experiments were run; one at 60 C with AIBN in diluted concentration (HS157-2) and the other at 75 C with benzoyl peroxide as initiator since benzoyl peroxide decomposes at a slower rate than AIBN in diluted concentration (HS157-3). All three conditions polymerized MMA in controlled manner. The copolymerization of MMA and PDMS macromonomer were run under the same conditions as HS157-3 shown below.

#### 4A. RAFT Copolymerization of MMA and PDMS-Macromonomer (5 mole%) (75°C, in solution)

Exp. No.	MMA, g	p-Xylene, wt. ratio to MMA	Dithioester (CDB), mg	Catalyst	MMA/CDB/Catalyst mol ratio	Temp., C
HS156	2.5	1 : 1	23	AIBN	300 / 1 / 0.5	75
HS157-1	3.0	1 : 0.29	14	AIBN	639 / 1 / 0.56	60
HS157-2	3.0	1 : 1	30	AIBN	300 / 1 / 0.5	60
HS157-3	3.0	1 : 1	30	BzPO	300 / 1 / 0.5	75

CDB: 2-phenylprop-2-yl dithiobenzoate (cumyl dithiobenzoate)

AIBN: 2,2'-Azobisisobutyronitrile,  $k_d=0.8-1.2 \cdot 10^{-5}$  at 60C in Bz,  $9.5 \cdot 10^{-5}$  at 77C in xylene.

BzPO: Benzoyl peroxide,  $k_d=2.6 \cdot 10^{-5}$  at 75C in Bz.

In a typical RAFT copolymerization, a 25 mL Schlenk flask equipped with a stir bar and containing a PDMS-MA (1.00g) was evacuated overnight and then filled with nitrogen. The free radical initiator and addition-transfer fragmentation agent and deoxygenated MMA (0.80 g) were mixed in a separate 25 mL round bottom flask under nitrogen and dissolved in deoxygenated xylene (0.80 g). This solution was cannula-transferred to the Schlenk flask under a nitrogen flow. A 0.1 mL aliquot of the reaction mixture was removed and the flask was placed into a 75 °C oil bath under nitrogen. Periodically, 0.1 mL aliquots of the reaction mixture were removed for the kinetic and molecular weight analysis. The conditions are summarized below.

Exp. No.	MMA, g	p-Xylene, wt. ratio to MMA	Macromonomer (MM), g	Dithioester (CDB), mg	Catalyst	MMA/MM/CDB/Catalyst mol ratio	Temp., C
HS159	0.8	1 : 1	1.0	8.5	BzPO	285/15 / 1 / 0.5	75

Macromonomer: HS134,  $M_n=2370$ (DP32), PDI=1.25, F=1.0

CDB: 2-phenylprop-2-yl dithiobenzoate (cumyl dithiobenzoate)

BzPO: Benzoyl peroxide,  $k_d=2.6 \times 10^{-5}$  at 75C in Bz.

The GC analysis revealed that MMA was consumed at the similar rate ( $k_p^{MMA} = 4.7 \times 10^{-5}$ ) as RAFT homopolymerization of MMA ( $k_p^{MMA} = 5.5 \times 10^{-5}$ ). It seemed that in all of these RAFT experiments, the polymerization rate was somehow accelerated at the later stage. The kinetics plot well fit to the line of  $\ln([M]_0/[M])=kt^2$  rather than  $\ln([M]_0/[M])=kt$ . The toluene-GPC showed continuous increase of the amount and molecular weight of copolymer along with the continuous decrease in the concentration of macromonomer. From the toluene-GPC of the periodical samples, PDMS- macromonomer conversions were calculated and their negative exponential values were plotted against the reaction time. The reactivity ratio of MMA ( $r_{MMA}$ ) obtained from Jaaks plot was 1.64. The molecular weight estimated from THF-GPC increased according to the monomer conversion and showed a sudden jump at high conversion. The polydispersity remained lower than 1.3 until the conversion reached 68%.

*4B. RAFT Copolymerization of MMA and PDMS-Macromonomer, (60°C, in solution)*

The reaction was run at 60 °C in order to determine how the temperature affects the reactivity. If the relative ratio of the rate of the transfer reaction to monomer addition reaction can be changed by changing temperature a change in the reactivity ratio of the macromonomer may be detected.

5                      Conditions for RAFT copolymerization of MMA and PDMS-Macromonomer are summarized in the following table.

Exp. No.	MMA, g	p-Xylene, wt. ratio to MMA	Macromonomer (MM), g	Dithioester (CDB), mg	Catalyst	MMA/MM/CDB/Catalyst mol ratio	Temp., C
HS164	2.15	1 : 1	2.48	23.0	AIBN	285/15 / 1 / 0.5	60

Macromonomer: HS94, Mn=2200, PDI=1.18, F=1.0  
CDB: 2-phenylprop-2-yl dithiobenzoate (cumyl dithiobenzoate)

10                      In this example the reactivity ratio of MMA ( $r_{\text{MMA}}$ ) calculated by the Jaacks method was 1.70, therefore,  $1/r_{\text{MMA}}$  was 0.59. This value was very close to the case when the RAFT was done at 75 °C; example 11, and a reactivity ratio of  $1/r_{\text{MMA}}=0.61$ . Both RAFT copolymerizations yielded reactivity ratio's lower than that in ATRP ( $1/r_{\text{MMA}}=0.73$ ). This is probably due to the relatively faster monomer addition in the RAFT polymerization compared with ATRP.

#### 5. Conventional Radical Copolymerization Initiated by AIBN.

15                      A 25 mL Schlenk flask equipped with a stir bar containing a PDMS-macromonomer (Mn = 2200, PDI = 1.18, F = 1.0, 0.98 g, 0.44 mmol) was evacuated overnight. The flask was filled with nitrogen and evacuated again. The degassing/back-filling procedure was repeated three times, and finally the flask was filled with nitrogen. AIBN (3.6 mg, 0.022 mmol) was added to a separate 25 mL round bottom flask equipped with a rubber septa and a stir bar. This flask was purged with a nitrogen gas flow for more than half an hour. Deoxygenated p-xylene (0.97 mL) and deoxygenated MMA (0.84 g, 8.36 mmol) were added via syringes. After the AIBN dissolved, the clear solution was cannula-transferred to the Schlenk flask under a nitrogen flow. A 0.1 mL aliquot of the reaction mixture was removed and the flask was placed into a 75 °C oil bath under nitrogen. Periodically, 0.1 mL aliquots of the reaction mixture were removed for kinetic and molecular weight analysis. These samples were diluted with toluene to 30 wt%,

subjected to GC to measure MMA conversion, toluene-GPC for the macromonomer conversion, and the THF-GPC to calculate the molecular weight of the generated copolymer.

5A. *Conventional Radical Copolymerization Initiated by PDMS-macro-azo initiator.*

A 25 mL Schlenk flask with a stir bar, containing a PDMS-macromonomer ( $M_n = 2200$ , PDI = 1.18,  $F = 1.0$ , 1.00 g, 0.45 mmol) was evacuated overnight. A PDMS-macroazo initiator (0.23 g, 0.023 mmol of azo group) was placed in a separate 25 mL round bottom flask, equipped with a stir bar, and sealed with a rubber septa. This flask was also evacuated overnight. These two flasks were filled with nitrogen and evacuated again. After this degassing/back-filling procedure was repeated three times, both the flasks were filled with nitrogen. Deoxygenated p-xylene (1.41 mL) and deoxygenated MMA (0.87 g, 8.63 mmol) were added to the round bottom flask via syringes. After the PDMS-macroazo initiator dissolved, the clear solution was cannula-transferred to the Schlenk flask under a nitrogen flow. The polymerization was performed at 75 °C and periodical sampling and measurements were done in the same way as in the case of AIBN initiated reaction described above.

Exp. No.	MMA, g	p-Xylene, wt. ratio to MMA	PDMS-MM, g	MMA/MM/AIBN mol ratio	Temp., C	Time, hr	Yield, g	$M_n$	$M_w/M_n$
HS170	2.3	1:1.5	2.82	380/20/1	75	5.5	3.4	139,400	5.68

Macromonomer: HS134,  $M_n=2370$

Copolymer was isolated by reprecipitation (MeOH/Acetone).

$M_n$  and  $M_w/M_n$  were calculated from THF-GPC.

The incorporation of the monomers into the copolymer were followed and the reactivity ratio's determined from a Jaacks plot. In contrast to the controlled ATRP polymerization, the  $r_{MMA}$  in the conventional radical polymerization was high and the resulting copolymer was heterogeneously branched. In contrast to the controlled polymerizations where heterogeneous branching is seen as a distribution along a polymer chain in standard radical polymerization this heterogeneous distribution is seen among the polymer chains.

The polydispersity became quite large. The copolymer was isolated by precipitation (MeOH/acetone) and the amount of residual macromonomer was decreased to less than 3 wt %.

6. *Redox Initiated Copolymerization MMA and PDMS-Macromonomer with Small Molecule Initiator.*

A 25 mL round bottom flask equipped with a stir bar containing a PDMS-macromonomer ( $M_n = 2370$ ,  $PDI = 1.25$ ,  $F = 1.0$ , 1.04 g, 0.44 mmol) was evacuated overnight and back-filled with nitrogen. In a 25 mL Schlenk flask, were placed a stir bar, CuBr (4.2 mg, 0.029 mmol), copper ( $Cu^0$ ) powder (7.4 mg, 0.117 mmol), Me4-cyclam (7.5 mg, 0.029 mmol). After this Schlenk flask was purged with nitrogen gas for half an hour, deoxygenated MMA (0.83 g, 8.3 mmol) was added and the solution was stirred under nitrogen for half an hour. Deoxygenated xylene (0.96 mL) was added into the round bottom flask and the PDMS-macromonomer was dissolved. This macromonomer solution was cannula-transferred to the Schlenk flask under a nitrogen flow. Ethyl 2-bromoisobutyrate (4.3  $\mu$  L, 0.029 mmol) was added to the flask through the capped arm on the flask. The polymerization was performed at 75 °C and periodic sampling and measurements were done in the same way as in the case of the ATRP reaction described above. In this system, the rate of deactivation was quite slow so that the reaction proceeds similar manner to the conventional radical polymerization. By Jaack's plot, the reactivity ratios of MMA and PDMS-macromonomer were 2.58 and 0.39 respectively.

6A. *Redox Initiated Copolymerization of MMA and PDMS-Macromonomer with PDMS-Macroinitiator*

MMA and PDMS- macromonomer were copolymerized in xylene with use of CuBr as a catalyst, a PDMS-MI having 2-bromoisobutyrate terminal group as an initiator, and cyclam as a ligand. The results are shown in Table 3.

Exp. No.	MMA, g	p-Xylene, wt. ratio to MMA	Macromonomer (SiM), g	Macroinitiator (MI), g	M/SiM/MI/CuBr/Ligand/Cu <sup>0</sup> mol ratio	Temp., C
HS146	0.82	1 : 1	0.99	0.43	285 / 15 / 1 / 1 / 1 / 5	75

Macromonomer: HS134, Mn=2370, DP=32, f=1.0

Macroinitiator: HS98, Mn=15590, Mw/Mn=1.10, f=0.95

Ligand: 1,4,8,11-tetramethyl-1,4,8,11-tetraazacyclotetradecane [cyclam]

Table 3. Redox initiated copolymerization of MMA and macromonomer

By the Jaack's plot, the reactivity ratios of MMA and PDMS-macromonomer were 1.90 and 0.53 respectively. Comparing to the case of redox polymerization with a low molecular weight initiator (ethyl 2-bromoisobutyrate), incorporation of PDMS-macromonomer to the copolymer was accelerated.

### 7. Summary of PDMS-macromonomer Copolymerization with MMA

The relative reactivity ratios of MMA and PDMS-macromonomer in various radical based polymerization systems are summarized in Table 4.

Table 4. Polymerization processes, reaction conditions and reactivity ratios.

Polymerization type		Exp. No.	Initial polymer concentration, wt%	Initial weight ratio MMA/SiM	[MMA] <sub>0</sub> , mol/L	[MM] <sub>0</sub> , mmol/L	$\frac{[MMA]_0}{[SiM]_0}$	Temp., C	r <sub>1</sub> <sup>a</sup>	1/r <sub>1</sub> <sup>a</sup>
ATRP	Solution	HS119	37	0.85	2.9	155	19.0	90	1.24	0.81
	Solution	HS160	38	0.80	2.9	151	19.1	75	1.37	0.73
	bulk	HS117	51	0.86	4.3	223	19.2	90	2.05	0.49
ATRP/MI	Solution	HS130	37	0.86	2.0	104	18.9	90	1.36	0.74
	bulk	HS120	62	0.83	3.3	180	18.7	90	1.17	0.85
	bulk	HS124	61	0.86	3.4	180	19.0	75	1.62	0.62
RP	Solution	HS110	37	0.86	2.9	155	18.9	75	3.07	0.33
RP/MI	Solution	HS128	37	0.87	2.4	128	19.0	75	2.37 <sup>b</sup>	0.42 <sup>b</sup>
Redox	Solution	HS138	38	0.81	2.9	152	19.0	75	2.58	0.39
Redox/MI	Solution	HS146	46	0.83	2.4	128	19.0	75	1.90	0.53
RAFT	Solution	HS159	38	0.80	2.9	151	19.0	75	1.64	0.61
	Solution	HS164	37	0.86	3.4	156	21.7	60	1.70	0.59
<hr/>										
Muller et al.										
MMA/PMMA(Pn=15)		no.38	2.0	0.8	0.12	9.7	12.4	60	1.53	0.65
RP		no.19	5.2	0.6	0.23	26	8.8	60	1.55	0.65
		no.24	10.1	0.6	0.46	52	8.8	60	2.14	0.47
		no.20	24.8	0.6	1.16	133	8.7	60	2.04	0.49
		no.21	38.2	0.6	1.89	214	8.8	60		
		no.39	2.0	6.9	1.02	10	102	60	1.12	0.89
		no.44	5.0	7.3	2.7	25	108	60	1.23	0.81

<sup>a</sup> Determined by Jaack's plot.

<sup>b</sup> Average value.

[MMA]<sub>0</sub>: 9.35 mol/L (MMA homopolymerization in bulk)

4.5 mol/L (MMA/MM(95/5) copolymerization in bulk)

As a whole, these results can generally be summarized as follows:

1.) In a controlled polymerization process such as ATRP the incorporation of the macromonomer into the polymer was greatly improved compared to a conventional radical polymerization, viz. the higher reactivity ratio of macromonomer, because the slow propagation rate allows one to ignore, or reduce the effects of diffusion control.

2.) Use of a compatible macroinitiator with the macromonomer can accelerate the incorporation of macromonomer into the copolymer in all radical polymerization processes. However, in the case where the propagation rate is high and the diffusion control can't be ignored the beneficial effect of use of a macroinitiator increased the reactivity of the macromonomer to a smaller extent.

3.) The beneficial effect of use of a compatible macroinitiator in ATRP was obvious when the polymerization was done at higher concentration of macromonomer because the growing chain and macromonomer have good compatibility each other.

The reduced reactivity of macromonomer will affect the final copolymer structure in different ways in RAFT and in the conventional free radical systems. In a RAFT process, all chains propagate simultaneously, therefore, every polymer chain will have similar branching structure although it may be intramolecularly heterogeneous. On the other hand, in the conventional radical process, the copolymer generated at the early stage of the polymerization has a totally different structure from that of the copolymer generated at the later stage. This phenomenon is due to the slow initiation throughout the reaction and progressive change in the feed composition as the comonomers are consumed at different rates. This leads to an intermolecularly heterogeneous structure for the copolymer although the degree of heterogeneity between the chains is less when a macroinitiator is employed.

In Table 5, the characterizations of PMMA-g-PDMS copolymer samples obtained so far are summarized.

Table 5. PMMA-g-PDMS copolymer samples for the study of properties.

Sample code	Polymerization type	Amount, g	$r_{MMA}$	Branch distribution		$M_n$	$M_w/M_n$	Mol% PDMS in copolymer MM, wt%	Residual MM, wt%
				Intramol	Intermol				
HS166	ATRP	3.2	1.34	homo	homo	86,020	1.58		1.3
HS164	RAFT	3.2	1.70	hetero	homo	101,600	2.59		2.7
HS170	Conv. radical	3.4	2.00	hetero	hetero	139,400	5.69		2.9

ATRP can be used to prepare homogeneously branched graft copolymers. The copolymers prepared using a conventional free radical method were heterogeneous in terms of copolymer composition, branching regularity, and molecular weights. However, control of concentration of macromonomers and other monomers during the polymerization and selection of the reactive end groups could produce more homogeneously branched polymers by conventional free radical polymerization. Graft copolymers obtained by ATRP (in both solution and semi-bulk) have predetermined molecular weights with much lower polydispersities than obtained in the conventional radical systems. In summary, the combination of ATRP and a macroinitiator that is soluble in the macromonomer is an effective way to control the graft copolymer structure in terms of main chain length/polydispersity and branch distribution/homogeneity.

RAFT also prepares a homogeneous graft copolymer but one with a more heterogeneous distribution of grafts along the backbone. Such a distribution can be considered a gradient copolymer between the low molecular weight monomer and the macromonomer. RAFT polymerizations may also prepare homogeneously distributed grafts by modification of the reactive end groups of the macromonomers using blends of macromonomers or controlling the concentrations of macromonomers in the polymerization process.

#### 8. Synthesis of PLLA-macromonomer and its copolymerization with MMA.

**Materials.** Tetrahydrofuran (THF) was distilled from purple sodium benzophenone ketyl solutions. Methyl methacrylate (MMA) (99%, Acros) was washed three times with 5% aqueous sodium hydroxide and once with water. After drying with magnesium sulfate, MMA was distilled from calcium chloride. 2-Hydroxyethyl methacrylate (HEMA) (97%, Aldrich) and 2-hydroxyethyl acrylate (HEA) (96%, Aldrich) were both dried over



molecular sieves and distilled. MMA and p-xylene were deoxygenated by bubbling nitrogen gas through them for more than one hour just before the polymerization. High purity L-lactide (L-LTD) (>99.9%) and D-lactide (D-LTD) (>99.9%) were supplied by Mitsui Chemicals and purified by recrystallization from toluene and dried in vacuum (<3 mmHg) overnight at room temperature. Copper chloride (CuCl) (98%, Aldrich) was stirred in glacial acetic acid overnight, filtered, and washed with absolute ethanol and ethyl ether under nitrogen. The solid was dried under vacuum at room temperature overnight. 4,4'-Di-n-nonyl-2,2'-bipyridine (dnNbpy) was synthesized by a modified literature procedure. Ethyl 2-bromoisobutyrate (EBiB) (98% Aldrich), benzoyl peroxide (BPO) (Fisher, 75%), stannous octoate (SnOct) (Alfa Aesar, technical grade) and all other reagents were used as received.

*Measurements.* Gel permeation chromatography (GPC) measurements in THF were conducted using a Waters 515 liquid chromatograph pump (1 mL/min, 30 °C) equipped with four columns (guard, 10 5 Å, 10 3 Å, 100 Å; Polymer Standards Service) in series with a Waters 2410 differential refractometer with diphenyl ether as an internal standard. The molecular weights of the copolymers were determined based on low polydispersity poly(methyl methacrylate) (PMMA) standards. The conversion of MMA was measured on a Shimadzu GC-14A gas chromatograph (GC) equipped with a FID detector using a widebore capillary column (30 m, DB-Wax, J&W Sci.). Injector and detector temperature: 250 °C; column temperature: 40 °C for 2 minutes followed by an increase to 160 °C at the rate of 40 °C/min and held for 2 minutes. The <sup>1</sup>H-NMR in deuterated chloroform (CDCl<sub>3</sub>) was measured with a 300 MHz Bruker spectrometer using Tecmag data acquisition software.

*Syntheses.* PLA macromonomers were prepared by a ring opening polymerization of lactide using 2-hydroxyethyl methacrylate or 2-hydroxymethyl acrylate as the initiator. Lactide is often contaminated by a small amount of hydroxy-compounds such as lactic acid, lactic acid dimer, and water.<sup>31</sup> It is extremely important to use highly purified lactide for the

macromonomer preparation because these impurities can act as undesirable initiators in the lactide polymerization and produce non-functionalized PLA and decrease the functionality.

*8A. ATRP of MMA in the presence of stannous octoate.*

5 Since a polylactic acid macromonomer with a methyl methacrylate end group (PLA-macromonomer), prepared by the ring-opening polymerization of lactide using stannous octoate (SnOct) as a catalyst, may contain a certain amount of Sn catalyst residue, the effect of tin on an ATRP polymerization was determined. Under optimal conditions for lactide polymerization, 0.08 wt% SnOct  
10 may be used. If this crude macromonomer was used at 5 mol% PLA-macromonomer for copolymerization with MMA, a maximum of 7.5 mol% of SnOct per MMA could be contained in the reaction mixture. Therefore, MMA was polymerized by ATRP in the presence and in the absence of SnOct to determine whether SnOct will interfere the ATRP. The results are shown in Table 6.

15 Table 6. ATRP of MMA.

Exp. No.	MMA, g(mmol)	Initiator	Xylene, g	MMA / I / CuCl / Ligand <sup>b</sup> mol ratio	Sn Oct, <sup>c</sup> mol % to Cu	Temp., C	Time, hr	Conversion <sup>d</sup> , %	M <sub>n</sub> <sup>e</sup>	M <sub>w</sub> /M <sub>n</sub> <sup>e</sup>
HS162-1	2.0(20)	EBiB <sup>a</sup>	2.0	300 / 1 / 1 / 2		90	6.4	49	18,000	1.13
HS162-2	2.0(20)	EBiB <sup>a</sup>	2.0	300 / 1 / 1 / 2	7.5	90	6.3	64	21,600	1.09

<sup>a</sup> EBiB: ethyl 2-bromoisobutyrate.

<sup>b</sup> Ligand: 4,4'-di(n-nonyl)-2,2'-bipyridine [dnNBPy].

<sup>c</sup> Stannous octoate.

<sup>d</sup> Determined by GC

<sup>e</sup> Determined by GPC (THF, PMMA standard).

Addition of SnOct to an ATRP polymerization of MMA seemed to be accelerate the polymerization rate and provide a polymer with narrower MWD; however his observation could be within experimental error. The molecular weight  
20 increase was linear and changes for both cases were similar. In conclusion, it is suggested that SnOct does not interfere significantly the ATRP.

*8B. Preparation of M-PLLA-macromonomer*

*Methacrylate-terminated Poly(L-lactic acid) Macromonomer (M-PLLA)(Run HS-168)*

L-LTD (20.0 g, 0.14 mol) was placed into a 100 mL round bottom flask equipped with a stir bar. After a toluene solution of SnOct (16 mg/mL-toluene) was added, the flask was capped with a rubber septa and vacuumed (1 mmHg) through a needle for more than three hours. The flask was filled with nitrogen gas, and HEMA (0.9 g, 5 mol %/LTD) was added via a syringe. The reaction flask was immersed in an oil bath of 120 °C and L-LTD was melted with stirring. Although the reaction mixture solidified in about one hour, the reaction was continued for three hours. After cooling, the reaction mixture was dissolved in chloroform (40 mL) and poured into 2-propanol (800 mL). The precipitated polymer was recovered by filtration and washed with cold methanol and dried in vacuum at room temperature. Yield = 18.6 g (89 %). The functionality ( $F = 1.0$ ) was calculated from the NMR peak intensities at  $\delta = 1.9$  ppm (s, 3H, methyl) for the methacrylate end-group and at  $\delta = 4.4$  ppm (q, 1H, methine) for the 2-hydroxypropionate end-group. The number average molecular weight ( $M_n = 2800$ ) was estimated from the peak intensities at  $\delta = 5.2$  ppm for the ester methine protons in the polyester chain and at  $\delta = 4.4$  ppm (q, 1H, methine) for the 2-hydroxypropionate end-group.  $M_w/M_n = 1.16$  (by GPC).

*8C. Methacrylate-terminated Poly(D,L-lactic acid) Macromonomer (M-PDLLA)(HS177-2)*

An amorphous M-PDLLA was prepared by the same procedure as M-PLLA except using a mixture of L-LTD (10 g, 0.07 mol) and D-LTD (10 g, 0.07 mol) as the monomer. The polymerization was performed at 130 °C for 2.5 hr. Yield = 18.7 g (89 %).  $F = 0.99$  (by  $^1\text{H-NMR}$ ).  $M_n = 3350$  (by  $^1\text{H-NMR}$ ).  $M_w/M_n = 1.20$  (by GPC).

*8D. Acrylate-terminated Poly(L-lactic acid) Macromonomer (A-PLLA)(HS177-3)*

A-PLLA was prepared by replacing HEMA with HEA in the method for M-PLLA synthesis. The polymerization was performed at 120 °C for 2.5 hr. Yield = 19.9 g (96 %).  $F = 0.98$  (by  $^1\text{H-NMR}$ ).  $M_n = 2690$  (by  $^1\text{H-NMR}$ ).  $M_w/M_n = 1.22$  (by GPC).

Table 7. Summary of ring-opening polymerizations of lactide (LTD).

Exp. No.	L-LTD, g(mol)	D-LTD	Initiator, mol%	SnOct, mol%	Temp., C	Time, hr	Yield, g(%)	M <sub>n</sub> (by NMR)	M <sub>n</sub> (by GPC)	M <sub>w</sub> /M <sub>n</sub>
HS168	20(0.14)		HEMA 5.2	0.028	120	3	18.6(89)	2,800	4,580	1.16
HS177-1	10(0.07)		HEMA 10.7	0.028	120	2.5	7.3(67)	1,770	2,500	1.13
HS177-2	10(0.07)	10(0.07)	HEMA 5.0	0.028	130	2.5	18.7(89)	3,350	4,737	1.20
HS177-3	20(0.14)		HEA 5.5	0.028	120	2.5	19.9(96)	2,690	5,101	1.22

HEMA: 2-hydroxyethyl methacrylate, HEA: 2-hydroxy acrylate.

SnOct: stannous octoate.

Polymer was isolated by reprecipitation (2-propanol/chloroform).

M<sub>n</sub> was evaluated by both H-NMR spectroscopy and THF-GPC.

M<sub>w</sub>/M<sub>n</sub> was calculated by THF-GPC.

Highly functionalized, low polydispersity PLLA-macromonomers were obtained. Although the molecular weight from GPC was higher than the value from NMR, it is consistent with our knowledge (M<sub>n</sub> from GPC is about 1.5 times higher than actual molecular weight). This macromonomer was obtained as a white powder, soluble in chloroform, THF, hot toluene, hot xylene.

Table 8. Characterization and solubility of the PLA macromonomers

Exp. No.	Terminol group	D/L form	Crystallinity	M <sub>n</sub> (NMR)	Solubility							
					CHCl <sub>3</sub>	THF	Toluene	Xylene	Acetone	2-Propanol	Methanol	n-BA
HS168	methacryloyl	L	Crystalline	2,800	s	s	sg	sg	ps	i	i	
HS177-1	methacryloyl	L	Crystalline	1,770	s	s	sg	sg	s	i	ps	sg
HS177-2	methacryloyl	DL	Amorphous	3,350	s	s	s	s	s	i	ps	s
HS177-3	acryloyl	L	Crystalline	2,690	s	s	sg	sg	ps	i	i	

Solubility; s: soluble, i: insoluble, ps: partially soluble, sg: soluble when hot but swollen gel when cold.

Sample HS177-1 was prepared to improve the solubility of the macromonomer and reduce the viscosity in the copolymerization mixture. HS177-2 was prepared in order to determine if the solubility and crystallinity had an affect on the macromonomer reactivity. HS177-3 was initially prepared for its copolymerization with acrylate monomer such as butyl acrylate. All of the macromonomers were obtained with predetermined molecular weights and narrow polydispersities. The methacryloyl terminated PLLA macromonomers will be called as M-PLLA-macromonomer and acryloyl terminated PLLA macromonomer will be called as A-PLLA-macromonomer. Poly(L-lactide) will be called PLLA, poly(DL-lactide) will be called PDLLA, and poly(D-lactide) will be called PDLA.

*GPC study:* Similar to the study of PDMS-macromonomer, a GPC study was done for the PLLA-macromonomers. THF solutions of the mixture of PLLA-macromonomer and PMMA with their various compositions were prepared and subjected to the THF-GPC measurement. Polymer composition and their peak intensities (area) showed a good linearity. The peak intensity ratio of PMMA to PLLA was found to be 2.17. THF solutions of the mixture of PLLA-macromonomer and PBA with their various compositions were prepared and subjected to THF-GPC measurement. Polymer composition and their peak intensities (area) showed a good linearity. The peak intensity ratio of PBA to PLLA was found to be 1.29.

#### *8E. Copolymerization of M-PLLA-macromonomer and MMA using ATRP*

ATRP copolymerization of M-PLLA-macromonomer and MMA was performed at 90 C. The concentration of M-PLLA-macromonomer in the feed was 3.5 mol %, or 50 wt %. The target degree of polymerization was 300. Initially, diphenyl ether (wt ratio to MMA: 1.7) was used for the solvent, but the M-PLLA-macromonomer solution became a gel at 90 C. Therefore, xylene (wt ratio to MMA: 1.0) was added to the flask via a cannula to form a solution which seemed to be clear and homogeneous at 90 C. MMA containing CuCl/dNbpy complex and ethyl bromoisobutyrate was added to the solution. The reaction mixture was brown homogeneous all the time.

Exp. No.	MMA, g	PLA-MM, g	p-Xylene, wt ratio to MMA	Diphenyl ether, wt ratio to MMA	M/MM/I/CuCl/dNbpy mol ratio	Temp., C
HS171	1.38	1.4	1.38	2.37	289/11/1/1/2	90

Macromonomer: HS168, Mn=2800

Initiator: ethyl 2-bromoisobutyrate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

#### Conditions for ATRP copolymerization of MMA and M-PLLA macromonomer

The kinetic samples, periodically taken from the reaction mixture, were diluted in THF and subjected to the GC and GPC for the estimation of the conversion of MMA and M-PLLA-macromonomer, respectively. Interestingly, the periodical sample taken at the early stage of polymerization were solidified easily at room temperature and a little hard to be dissolved in THF, while the samples at

the later stage were easier to be dissolved in THF. In THF GPC, a continuous decrease of M-PLLA-macromonomer and a continuous increase of copolymer were observed. Rather surprisingly, the conversion rate of M-PLLA-macromonomer was higher than that of MMA and the reactivity ratio of MMA ( $r_{\text{MMA}}$ ) was 0.58.

$$r_{\text{MMA}} = \ln([MMA]_0/[MMA]) / \ln([\text{macromonomer}]_0/[\text{macromonomer}])$$

The relative reactivity of the macromonomers was evaluated by  $1/r_{\text{MMA}} = 1.75$ . This value means that the rate constant for the reaction of the methacrylate radical at the growing chain with M-PLLA is 1.75 times higher than that with MMA.

#### 8F. Conventional radical polymerization of M-PLLA-macromonomer with MMA

In the above example the copolymerization of MMA with M-PLLA-macromonomer resulted in the faster incorporation of MM than MMA into copolymer, which is rather unusual in the copolymerization of macromonomers with low molecular weight comonomers. For a comparison, a conventional radical copolymerization of M-PLLA-macromonomer with MMA using benzoyl peroxide as a initiator under the same conditions.

Table 9. Conventional radical polymerization of M-PLLA-macromonomer with MMA

Exp. No.	MMA, g	M-PLLA-MM, g	p-Xylene, g	Diphenyl ether, g	M/MM/I mol ratio	Temp., C
HS175	1.38	1.4	1.38	2.37	289/11/1	90

Macromonomer: HS168,  $M_n=2800$

Initiator: benzoyl peroxide.  $k_d$  at 90 C:  $1.0-1.9 \times 10^{-4}$ .

Examination of the kinetics and Jaacks plot showed that MMA and M-PLLA-macromonomer were incorporated into copolymer at a similar rate. The reactivity ratio of MMA ( $r_{\text{MMA}}$ ) was 1.09. The molecular weight changes showed a typical behavior of the conventional radical polymerization. Jose L. Eguiburu, Jose F-Berridi, Julio S. Roman, *Polymer* 1996, 37(16), 3615; reported the free radical polymerization of M-PLLA-macromonomer ( $M_n=4500$ ) with MMA and MA using AIBN as the initiator in dioxane at 60° C. They found  $r_{\text{MMA}}=1.01 \pm 0.17$  for the

copolymerization of MMA with M-PLLA-macromonomer, and  $r_{\text{MMA}}=0.85\pm0.01$  for the copolymerization of MMA with 2-acetoxyethyl methacrylate (a low molecular weight homologue of the M-PLLA-macromonomer). In the article they assumed that the reactivity ratio of the macromonomer was similar to that of the model compound and that both polymers are miscible. We believe that the difference between the two values is meaningful. Even if we can ignore any incompatibility effect on the macromonomer reactivity, there still exist the diffusion control effect. Therefore, we propose that the inherent reactivity of the terminal group of the macromonomer, the acylethyl methacrylate group, is higher than MMA and the overall reactivity of the macromonomer was reduced by the diffusion effect and coincidentally became similar to that of MMA ( $1/r_{\text{MMA}}=1.0$ ). In contrast to a controlled radical polymerization, in a standard free radical copolymerization the polydispersity of the resulting copolymer may be broad, (above 2.0).

#### 8G. ATRP of M-PDLLA-macromonomer with MMA

##### Reaction Conditions:

Exp. No.	MMA, g	M-PDLLA-MM, g	p-Xylene, g	Diphenyl ether, g	M/MM/I/CuCl/dNbpy mol ratio	Temp., C
HS191	1.74	2.11	1.74	2.99	289/11/1/1/2	90

Macromonomer(M-PDLLA-MM): HS177-2,  $M_n=3350$

Initiator: ethyl 2-bromoisobutyrate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

This is the experiment was designed to determine if the solubility of the PLA macromonomer has an influence its reactivity. Unlike M-PLLA, M-PDLLA showed good solubility in various solvents and was soluble in the reaction solvent even at room temperature. The GPC traces of the kinetic samples were similar to Figure 2, and the conversion of the macromonomer was calculated in the same way. (The intensity ratio of PMMA to PDLLA at the same concentration was IR = 1.91). The reactivity ratio  $r_{\text{MMA}}$  was  $0.68 \pm 0.01$ , which is very close to the  $r_{\text{MMA}}$  in the copolymerization of M-PLLA-macromonomer and MMA ( $r_{\text{MMA}}=0.67$ ). This gives a value for  $1/r_{\text{MMA}} = 1.47$ ; which indicates the solubility of PLA macromonomer does not significantly affect the copolymerization reactivity as long as the reaction mixture is homogeneous.

*8H. Preparation of homogeneously branched PMMA-g-PLLA graft copolymer by ATRP*

In the copolymerization of MMA and PLL-macromonomer, it has been revealed that methacrylate-terminated PLLA macromonomer (M-PLLA-macromonomer) was copolymerized faster than MMA whereas acrylate-terminated PLLA-macromonomer (A-PLLA-macromonomer) was copolymerized slower than MMA. We noted that the relative reactivity ( $1/r_{\text{MMA}}$ ) of M-PLLA-macromonomer and A-PLLA-macromonomer to MMA was 1.4 – 1.8 and 0.6, respectively, and the multiplication of both value is coincidentally close to one.

Therefore we believed that if we used a 1:1 mixture of M-PLLA-macromonomer and A-PLLA-macromonomer, we could expect that M-PLLA-macromonomer will be incorporated into the copolymer at the early stage and A-PLLA-macromonomer will be incorporated at the later stage in the copolymerization and an homogeneously branched graft copolymer will be generated.

Table 10. ATRP copolymerization of MMA with a mixture of M-PLLA/A-PLLA-macromonomer.

Exp. No.	MMA, g	M-PLLA-MM, g	A-PLLA-MM, g	p-Xylene, g	Diphenyl ether, g	M/MM/I/CuCl/dNbpy mol ratio	Temp., C
HS192	2.76	1.40	1.35	2.76	4.74	289/11/1/1/2	90

M-PLLA-MM: HS168,  $M_n=2800$ ,  $M_w/M_n=1.16$

A-PLLA-MM: HS177-3,  $M_n=2690$ ,  $M_w/M_n=1.22$

Initiator: ethyl 2-bromoisobutyrate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

This time, as expected, MMA and PLLA-macromonomer were incorporated into copolymer at a similar rate. The reactivity ratio of MMA ( $r_{\text{MMA}}$ ) was 1.07. This means that one could get the PMMA-g-PLLA graft copolymer with not only intermolecularly but also intramolecularly homogeneous branching. The molecular weight of the resulting obtained copolymer was close to the predetermined value ( $M_n=60000$ ). The polydispersity was quite low unlike the material prepared by a standard free radical copolymerization.  $M_w/M_n = 1.15$  (by GPC).



The reactivity ratios obtained in this series of examples and published in the literature are summarized in Table 11. One should not directly compare the reactivity ratios among the polymerization systems performed under different conditions (temperature, solvent, concentration) but the data can be used for relative comparison. It should be noted that the reactivity ratio of MMA for ATR copolymerization of MMA and HEMA was comparable to the value ( $r_{\text{MMA}} = 0.75$ ) reported by Fink for conventional radical copolymerization of these two monomers in bulk at 80 °C by (Fink, J. K. *Makromol. Chem.* 1981, 182, 2105-2107)

Table 11. Reactivity ratios for the copolymerization of MMA ( $M_1$ ) with macromonomer comonomers ( $M_2$ )

$M_2$	Polymerization system	Solvent <sup>h</sup>	$[M_1]_0$	Temp.,	$r_1^i$	$1/r_1$
M-PLLA <sup>a</sup>	ATRP	DPE/xylene	2	90	0.57 ±0.02	1.75
M-PDLLA <sup>b</sup>	ATRP	DPE/xylene	2	90	0.68 ±0.17	1.47
HEMA <sup>c</sup>	ATRP	Xylene	4	90	0.67 ±0.02	1.49
HEMA <sup>c</sup>	Conventional (AIBN)	(bulk)	9	80	0.75	1.33
AEM <sup>d</sup>	Conventional (AIBN)	Dioxane	1	60	0.85 ±0.01	1.18
M-PLLA <sup>e</sup>	Conventional (AIBN)	Dioxane	1	60	1.01 ±0.17	0.99
M-PLLA <sup>a</sup>	Conventional (BPO)	DPE/xylene	2	90	1.09 ±0.05	0.91
A-PLLA <sup>f</sup>	ATRP	DPE/xylene	2	90	1.63 ±0.10	0.61
HEA <sup>g</sup>	ATRP	Xylene	2	90	1.57 ±0.07	0.64

5

<sup>a</sup>Methacrylate-terminated poly(L-lactic acid) macromonomer,  $M_n = 2800$ .<sup>b</sup>Methacrylate-terminated poly(D,L-lactic acid) macromonomer,  $M_n = 3350$ .<sup>c</sup>2-Hydroxyethyl methacrylate.<sup>d</sup>2-Acetoxyethyl methacrylate.10 <sup>e</sup>Methacrylate-terminated poly(L-lactic acid) macromonomer,  $M_n = 4500$ .<sup>f</sup>Acrylate-terminated poly(L-lactic acid) macromonomer,  $M_n = 2690$ .<sup>g</sup>2-Hydroxyethyl acrylate.<sup>h</sup>DPE: diphenyl ether.<sup>i</sup>Determined by Jaacks method.

15

8I. *Copolymerization of A-PLLA- macromonomer and n-butyl acrylate using ATRP*

Table 12. ATRP copolymerization of n-butyl acrylate and A-PLLA-

Exp. No.	n-BA g	A-PLLA-MM, g	Toluene, g	M/MM/I/CuBr/dNBpy mol ratio	Temp., C	Time, hr	Conversion, %	
							BA	MM
HS185	3.53	2.69	5.3	289/11/1/1/2	90	7.2	7	2

Macromonomer(A-PLLA-MM): HS177-3, Mn=2690

Initiator: methyl 2-bromopropionate

Ligand: 4,4'-di(n-nonyl)2,2'-bipyridine [dnNBPy]

macromonomer.

5

This material is an interesting graft copolymer since the polymer has a soft segment backbone and biodegradable hard segments which have some ability of aggregation (crystallization). It may be useful as biodegradable thermoplastic elastomeric materials. This experiment was just the first attempt to make

10

polyacrylate-g-PLLA copolymer because the poor solubility of the PLLA in toluene, a large excess of solvent was employed and that made the reaction very slow. One would recommend the use a more active catalyst system using either triamine or picolylamine as ligand.

8J. *Copolymerization of A-PLLA- macromonomer and n-butyl acrylate using*

15

*ATRP with Picolylamine as the ligand.*

Table 13. ATRP of n-BA .

Exp. No.	n-BA, g(mmol)	BA / I <sup>a</sup> / CuBr / CuBr <sub>2</sub> / Ligand <sup>b</sup> mol ratio	Solvent, <sup>c</sup> g	Temp., °C	Time, hr	Conversion <sup>d</sup> , %	M <sub>n</sub> <sup>e</sup>	M <sub>w</sub> /M <sub>n</sub> <sup>e</sup>
HS203-1	3.85(30)	300 / 1 / 0.5 / 0.025 / 0.525	Toluenc 11.6	90	22.3	11.4	3,887	1.22
HS203-2	3.85(30)	300 / 1 / 0.5 / 0.025 / 0.525	DMF 11.6	60	22.3	57.3	11,808	1.38

<sup>a</sup> Initiator: Methyl 2-bromopropionate.

<sup>b</sup> Ligand: N,N-Bis(2-pyridylmethyl)octylamine [BPMOA].

<sup>c</sup> 0.4 ml of anisol was added as the internal standard.

<sup>d</sup> Determined by GC.

<sup>e</sup> Determined by GPC (THF, PS standard).

The reaction in toluene was very slow probably due to the poor solubility of catalyst. The reaction mixture was a little turbid yellow however when

DMF was used as solvent the reaction mixture was clear yellow, and polymerization did occur

9. *Block copolymers wherein each block comprises a graft copolymer*

Since some of the examples of graft-copolymers described above were prepared by ATRP they retain at the growing chain end a radically transferable atom and are therefore still active macroinitiators for further CRP (co)polymerizations. The synthesis of a second block, also comprising a graft copolymer; e.g. PMMA grafted through PDMS, was now prepared by ATRP.

9A1. *(PMMA-g-PLLA)-b-(PMMA-g-PDMS) block copolymer*

The graft-copolymer, PMMA grafted with PLLA, prepared as above, was employed as a macroinitiator for the copolymerization of MMA with a PDMS macromonomer. The targeted molecular weight for the MMA backbone of the second block is  $15000 \text{ g.mol}^{-1}$  and the macromonomer PDMS had a  $M_n = 3000 \text{ g.mol}^{-1}$ . The amounts of each component of this reaction are summarized in the following Table 14.

Table 14.

Methyl methacrylate	0.255 ml	$2.39 \cdot 10^{-3} \text{ mol}$
Macromonomer PDMS	0.262 g	$8.73 \cdot 10^{-5} \text{ mol}$
Macroinitiator PMMA/PLLA	700 mg	$\sim 1.6 \cdot 10^{-5} \text{ mol}$
CuCl	2.5 mg	$2.52 \cdot 10^{-5} \text{ mol}$
d-n-bipy	20.4 mg	$4.99 \cdot 10^{-5} \text{ mol}$
p-xylene	0.7 ml	
Phenyl ether	0.7 ml	

*Experimental procedure:* The macroinitiator (powder) and the macromonomer (liquid) were added in a dry schlenk flask. 1mm Hg vacuum was applied for 30 minutes to removing oxygen. Then, degassed xylene and diphenyl ether were added. The flask was heated with a heat gun until macrocomponents were dissolved. A solution of CuCl and dn-bipy in MMA was added and the mixture was heated at  $90^\circ\text{C}$ . After 20 hours, the reaction was stopped. The resulting polymer was dissolved in 2ml of

THF then precipitated in cold methanol, washed two times with methanol and dried under 1mm vacuum. 590mg of white powder was obtained.

*Analysis:*

*THF-GPC:* The average molecular weights and the polydispersity indices we obtained are summarized in Table 15.

Table 15

Time	Mn (g/mol)	Ip
0	43000	1.12
3	46700	1.14
6	49500	1.16
20	55500	1.19

Molecular weight increases versus time. We have undoubtedly chain extended our first polymer block.

<sup>1</sup>H-NMR: Analysis of the NMR spectra permits determination of the final composition of the polymer. The composition of the macroinitiator was 39% PMMA, 61% PLLA. The composition of the final polymer is 45% PMMA, 40% PLLA and 14% PDMS. As we have no GPC information concerning the consumption of the PDMS macromonomer, the PDMS signal of the NMR could be the remaining macromonomer. The macromonomer PDMS is soluble in hexane. So, the polymer was washed several times with hexane. After those washes, the NMR composition was nearly the same. This indicates that the PDMS macromonomer is incorporated in the polymer chains. So we have clearly extended the graft-macroinitiator with PMMA and incorporated the PDMS macromonomer forming a methyl methacrylate based block copolymer, one wherein one block is MMA grafted with PLLA and the second block is MMA grafted with PDMS.

*9A2 (PMMA-g-PDLA)-b-(PMMA-g-PDMS) block copolymer*

The macromonomer, and the first block of the block copolymer, PMMA-g-PDLA were prepared as described above. The molecular weight of the first block (macroinitiator) is 36800 g.mol<sup>-1</sup> (THF GPC measurement) and the molecular weight of the PDMS macromonomer is 3000 g.mol<sup>-1</sup> (<sup>1</sup>H NMR

measurement). The targeted degree of polymerization for the MMA backbone for the second block is 150. The targeted degree of polymerization for the macromonomer PDMS is 5. Therefore, the targeted molecular weight for the second block is  $30000 \text{ g.mol}^{-1}$ . The targeted molecular weight of the final polymer is consequently  $66800 \text{ g.mol}^{-1}$ . The amounts of each component of this reaction are summarized in Table 16.

Table 16

Methyl methacrylate	304 ml	$2.85 \cdot 10^{-3} \text{ mol}$
Macromonomer PDMS	285 mg	$9.50 \cdot 10^{-5} \text{ mol}$
Macroinitiator PMMA/PLLA	700 mg	$1.90 \cdot 10^{-5} \text{ mol}$
CuCl	4 mg	$4.04 \cdot 10^{-5} \text{ mol}$
d-n-bipy	32.9 mg	$8.08 \cdot 10^{-5} \text{ mol}$
p-xylene	0.7 ml	
Phenyl ether	0.7 ml	

*Experimental procedure:* The macroinitiator (powder) and the macromonomer (liquid) were added in a dry schlenk flask. 1mm Hg vacuum was applied for 30 minutes for removing oxygen. Then, degassed xylene and diphenyl ether were added. The flask was heated with a heat gun until macro-components were dissolved. A solution of CuCl and diN-bipy in MMA was added and the mixture was heated at  $90^{\circ}\text{C}$ . After 32 hours, the reaction was stopped. The resulting polymer was dissolved in 2ml of THF then precipitated in cold methanol, washed with methanol and hexanes and dried under vacuum. The average molecular weights and the polydispersities of the block copolymer were determined by THF-GPC based on PMMA standards calibration. Conversions of MMA and PDMS-MM were calculated from the area of the monomer pics.

Table 17

Time (h)	Mn (g.mol <sup>-1</sup> )	Ip	Conv. MMA	Conv. PDMS-MM
0	36800	1.13	0 %	0 %
4	49500	1.20	49 %	13 %
8	56000	1.22	68 %	39 %
21	58900	1.37	91 %	58 %
32	61900	1.34	90 %	70 %

First-order kinetics plots  $\ln([M]_0/[M])$  versus time obtained for conversion of MMA and the macromonomer PDMS-MM during this experiment. All these curves are linear. The polymerization of the second block is perfectly controlled by ATRP. However, the consumption of MMA is much faster than PDMS-MM. A Jaacks plot  $\ln([M]_0/[M])_{\text{MMA}}$  versus  $\ln([M]_0/[M])_{\text{PDMS-MM}}$  for this experiment was prepared and the slope of this Jaacks plot permits calculation of the reactivity ratio  $r_{\text{MMA}}$  for the copolymerization of MMA and PDMS macromonomer. We have calculated  $r_{\text{MMA}} = 2.7$ . This value is high and proves that the incorporation of the PDMS macromonomer in the polymer chain is not regular. This behavior is mostly due to the repulsive interaction between PMMA and PDMS and results in the formation of a gradient copolymer as the second block, a copolymer wherein the grafted macromonomer is concentrated at one end of the block, the end furthest from the first block.

It is also possible to prepare the block copolymer starting with a PMMA-g-PDMS macroinitiator and then copolymerizing MMA and PLLA-MM. These two different block copolymers have been synthesized: JFL023 where the first synthesized block was the PMMA-grafted PDLA block and JFL028 where the first synthesized block was the PMMA-grafted PDMS block. The reactivity ratios  $r_{\text{MMA/PDLA-MM}}$  and  $r_{\text{MMA/PDMS-MM}}$  were calculated for both copolymers, using the Jaacks plot method. Figure 3 shows these values and how these values impact the corresponding structures of the copolymers.

When the block PMMA-grafted PDLA is synthesized first (JFL023), the grafting segments are mostly located towards each the terminii of the

backbone (upper structure shown in Figure 3). I.e. it is almost an ABC triblock copolymer. The A block is PMMA-g-PDLA, the B block a pure PMMA segment, and the C block PMMA-g-PDMS. This results from consideration of the impact of the reactivity ratio  $r_{\text{MMA/PDLA-MM}} = 0.6$  in the first block, PMMA-grafted PDLA, consequently the grafted PDLA segments are mostly incorporated in the polymer chain at low MMA conversion. For the second block PMMA-grafted PDMS, due to the repulsive interaction between PMMA and PDMS, the reactivity ratio is really high  $r_{\text{MMA/PDMS-MM}} = 2.7$ , consequently the grafted PDMS segments are mostly incorporated in the polymer chain at high MMA conversion. In conclusion, although the overall structure of the copolymer is symmetric, the distribution of the branching on the backbone is not regular with each of the graft segments concentrated at the chain ends.

When the block PMMA-grafted PDMS is synthesized first (JFL028), the distribution of the grafting segments is more regular. For the first PMMA-grafted PDMS, the reactivity ratio  $r_{\text{MMA/PDMS-MM}} = 1.3$ . For the synthesis of the second block PMMA-grafted PDLA, there is no problem due to the repulsive effect and consequently the reactivity ratio is low  $r_{\text{MMA/PDLA-MM}} = 0.5$ . This block copolymer more closely resembles an AB block copolymer, (PMMA-g-PDMS)-*b*-(PMMA-g-PDLA); as represented by the lower structure in Figure 3.

This clearly shows that in the preparation of block copolymers wherein each block is a graft copolymer the order of synthesis of each block significantly impacts the structure or topology of the final product and the distribution of the macromonomers, or grafts, along the backbone polymer can be changed by order of block formation even when the block copolymers contain the same concentration of the same macromonomers and the block copolymer segments were prepared under the same conditions.

It is expected that with proper selection of macromonomers in such AB or ABC block-graft copolymers comprising different composition for the graft copolymer in each block the copolymer can phase separate into either a bi-phasic or a tri-phasic bulk morphology.

#### 10. Gradient Comonomer Copolymers

##### 10A: Synthesis of *abi*-graft gradient copolymer by ATRP



In this series of examples we will examine the effect of macromonomer end group reactivity and macromonomer composition in a copolymerization reaction involving the copolymerization of two macromonomers of different composition with a small molecular weight monomer.

5 The macromonomers PDLA-MM and PDMS-MM were synthesised as described above. The molecular weights of both macromonomers was 3000 g.mol<sup>-1</sup> (<sup>1</sup>H NMR measurement). The targeted degree of polymerization of the MMA backbone is 300. The targeted degree of polymerization for each macromonomer is 5. Therefore, the targeted molecular weight of the final polymer  
10 is 60000 g.mol<sup>-1</sup>. The amounts of each component of this reaction are summarized in Table 18.

*Experimental procedure:* PDLA macromonomer (powder) and PDMS macromonomer (liquid) were added to a dry schlenk flask. 1mm Hg vacuum was applied for 30 minutes for removal of oxygen. Then, degassed xylene and  
15 diphenyl ether were added. The flask was heated with a heat gun until macromonomers were dissolved. A solution of CuCl and diN-bipy in MMA was added. Then a solution of ethyl 2-bromoisobutyrate in MMA was added and the mixture was heated at 90°C. After 21 hours, the reaction was stopped. The resulting polymer was dissolved in 2ml of THF then precipitated in cold methanol,  
20 washed with methanol and hexanes and dried under vacuum.

Table 18

Methyl methacrylate	1.6 g	1.60 10 <sup>-2</sup> mol
Macromonomer PDMS	0.8 g	2.66 10 <sup>-4</sup> mol
Macromonomer PDLA	0.8 g	2.66 10 <sup>-4</sup> mol
Ethyl 2-bromoisobutyrate	7.8 µl	5.33 10 <sup>-5</sup> mol
CuCl	10.6 mg	10.66 10 <sup>-5</sup> mol
d-n-bipy	47.2 mg	2.12 10 <sup>-4</sup> mol
p-xylene	1.5 ml	
Phenyl ether	1.5 ml	

*Analysis:*

25 *THF-GPC:* The average molecular weights and the polydispersities of the dual gradient graft copolymer were determined by THF-GPC based on PMMA standards calibration. There are three different populations seen in

the GPC curves, due to diphenyl ether, PDLA macromonomer and the synthesized polymer. Conversions of PDLA-MM was calculated by comparing the areas of the corresponding signal.

*Toluene-GPC*: was also run and the chromatograms obtained show four different populations; due to MMA, diphenyl ether, PDMS macromonomer and the synthesized polymer. Conversion of PDMS-MM was calculated by comparing the areas of the corresponding signal.

GC: MMA conversion was calculated by comparing the MMA pick area and the internal standard (p-xylene) pick area. The results are given in Table

19.

Table 19

Time	Mn (g.mol <sup>-1</sup> )	Ip	Conv. MMA	Conv. PDMS-MM	Conv. PDLA-MM
1 h 30'	28800	1.15	36 %	10 %	42 %
3 h 20'	37700	1.14	50 %	30 %	56 %
5 h	41500	1.15	54 %	44 %	60 %
7 h 30'	45500	1.16	65 %	57 %	73 %
21 h	52200	1.20	-	69 %	95 %

Linear first-order kinetics plots  $\ln([M]_0/[M])$  versus time were obtained for MMA, PDLA-MM and PDMS-MM during this experiment. Jaacks plots for  $\ln([M]_0/[M])_{\text{MMA}}$  versus  $\ln([M]_0/[M])_{\text{PDLA-MM}}$  and  $\ln([M]_0/[M])_{\text{MMA}}$  versus  $\ln([M]_0/[M])_{\text{PDMS-MM}}$  were prepared and the slopes of these Jaacks plots permits to calculate the reactivity ratios  $r_{\text{MMA/PDMS-MM}} = 1.4$  and  $r_{\text{MMA/PDLA-MM}} = 0.8$ . These values prove that the synthesized copolymer is a gradient copolymer of both macromonomers. A graphical representation of the calculated structure of the dual macromonomer gradient graft copolymer is shown in Figure 4.

A second experiment was run, JFL031 where the gradient copolymer is initiated by a PDMS-Br (Mn ~ 8000 g.mol<sup>-1</sup>) macroinitiator and it was determined that when the gradient copolymer is initiated by such macroinitiator, the reactivity ratios were:  $r_{\text{MMA/PDLA-MM}} = 0.9$  and  $r_{\text{MMA/PDMS-MM}} = 1.75$ . This is not significantly different than when the small molecule initiator was used.

This result was initially unexpected but could be a further indicator of the necessity to consider the instantaneous composition of the growing backbone

chain when envisioning which macromonomer will be preferentially incorporated into the copolymer. The use of a PDMS macroinitiator has a significant effect in compatibilizing a low molecular weight growing MMA polymer chain with the PDMS macromonomer but this effect is essentially cancelled when a PDLA  
5 macromonomer is incorporated into the chain. One would expect that if the second co-macromonomer also had a high reactivity ratio with MMA then preferential incorporation of the macromonomer with a compatible macroinitiator would be seen.

*10B: Synthesis of a gradient copolymer by ATRP using a hybrid catalyst*

10 One concern with ATRP is removal of the catalyst from the final product and we have developed a hybrid catalyst system for the process; U.S. patent application 09/872,056, that significantly reduces the concentration of the transition metal in the final polymeric material. Residual transition metal is of particular concern when a multi-component system is being formed since one of  
15 the components could selectively interact with the transition metal to form a colored product.

*Grafting through copolymerization conditions :*

PLA macromonomer (5 eq.), PDMS macromonomer (5 eq.), p-xylene and phenyl ether were added in a dry schlenk flask. The mixture was  
20 thoroughly purged by nitrogen flushing then heated in order to obtain a perfect solubilization of the crystalline PLA macromonomer. Then, a dispersion of (CuBr/PS-dMBpy) (2 eq.) in methyl methacrylate (250 eq.) was added into the schlenk flask. A solution of (CuBr<sub>2</sub>(Me<sub>6</sub>TREN)) (0.01 eq.) in acetone was added via a degassed syringe. Finally, a solution of ethyl 2-bromoisobutyrate (1 eq.) in  
25 methyl methacrylate (50 eq.) was added. The mixture was thermostated at 90°C and samples were withdrawn through a degassed syringe at timed intervals.

*Analysis conditions:* The average molecular weights and the polydispersities of the graft copolymer were determined by THF-GPC based on PMMA standards calibration. PLA macromonomer conversion  
30 was calculated by THF-GPC by comparing the macromonomer pick area to the pick area of the diphenyl ether standard. PDMS macromonomer conversion was calculated by toluene-GPC by comparing the

macromonomer pick area to the pick area of the diphenyl ether standard. PLA macromonomer conversion was calculated by GC by comparing the monomer pick area to the pick area of diphenyl ether and p-xylene.

*Results and discussion:* An examination of the first-order kinetics plots  $\ln([M]_0/[M])$  versus time obtained for MMA, PDLA-MM and PDMS-MM during the copolymerization experiment showed that the kinetics of copolymerization is the same as that obtained with a homogeneous catalyst system; i.e. regardless of the catalyst system a controlled polymerization occurred. In the presence of the hybrid catalyst, the control of molecular weight is as good as in the presence of the homogeneous catalyst. However, polydispersity is slightly broader in the case of hybrid catalyst. These results prove that the hybrid catalyst system (CuBr/PS-dMBpy)/(CuBr<sub>2</sub>(Me<sub>6</sub>TREN)) allows one to control the synthesis of well-defined graft copolymer as efficiently as the classic homogeneous system (CuBr (d-n-Bipy)<sub>2</sub>).

*10C: Synthesis of a gradient copolymer by RAFT:*

A gradient copolymer was synthesized by RAFT. The amounts of each component of this reaction are summarized in Table 20.

Table 20

Methyl methacrylate	1.2 g	0,012 mol
Macromonomer PDMS	0.6 g	$2 \cdot 10^{-4}$ mol
Macromonomer PDLA	0.6 g	$2 \cdot 10^{-4}$ mol
Cumyl dithio benzoate	10.9 mg	$4 \cdot 10^{-5}$ mol
Benzoyl peroxide	2.6 mg	$1 \cdot 10^{-5}$ mol
p-xylene	1.2 ml	
Phenyl ether	1.2 ml	

*Experimental procedure:* The macromonomers were added in a dry schlenk flask. 1mm Hg vacuum was applied for 30 minutes for removing oxygen. Then, degassed xylene and diphenyl ether were added. The flask was heated with a heat gun until macrocomponents were dissolved. A solution of cumyl dithio

benzoate and benzoyl peroxide in MMA was added and the mixture was heated at 75°C. After 80 hours, the reaction was stopped. The obtained polymer was dissolved in 2ml of THF then precipitated in cold methanol, washed with methanol and hexanes and dried under vacuum. Table 21 shows the experimental results obtained for this experiment. The polymerization is extremely slow and, even after 3 days of reaction, the conversion of all monomers is still incomplete.

Experimental molecular weights are higher than expected but increase slowly with conversion. These results indicate that it is difficult, but still possible to achieve such complex bi-graft gradient structures by RAFT polymerization.

Table 21

Time	Mn (g.mol <sup>-1</sup> )	Ip	Conv. MMA	Conv. PDMS-MM	Conv. PDLA-MM
6 h	21500	1.09	15 %	0 %	0 %
21 h	49600	1.14	50 %	24 %	36 %
30 h	51400	1.14	-	24 %	36 %
46 h	51500	1.18	60 %	24 %	42 %
54 h	56600	1.25	60 %	24 %	55 %
80 h	56000	1.24	60 %	37 %	-

#### 11. *PMMA-b-Polypropylene and PMMA-g-Polypropylene Copolymer*

Two approaches were taken for the preparation of a polypropylene based macroinitiator for the ATRP copolymerization of a polypropylene macromonomer with methyl methacrylate. The first was the hydrosilation of a vinyl terminated polypropylene with chlorodimethyl silane followed by reduction of the Si-Cl bond to Si-H using LiAlH<sub>4</sub> and further reaction with allyl-2-bromoisobutyrate to form the macroinitiator. The second route is in many ways the reverse approach with the first step involving the reaction of allyl-2-bromoisobutyrate with tetramethyldisiloxane and subsequent reaction with the vinyl terminated polypropylene.

##### 11A. *Synthesis of a Polypropylene Macroinitiator (Route 1).*

##### 11A1. *Hydrosilation of the unsaturated chain end of PP*

Concentration ratio: olefin : silane : catalyst : ligand = 1 : 2 : 0.005 :

0.01

A 50 mL three-neck round bottom flask, previously dried in the oven for 3 h, was backfilled with nitrogen and charged with 806 mg (0.005 mmol) of methylnaphtoquinone. The Karstedt's catalyst in xylene solution (0.096 mol/L, 26 mL, 0.0025 mmol) was then added at once, leading to a gold homogeneous solution. 1.5 g (0.5 mmol) of vinyl-terminated polypropylene was dissolved into 3 mL of xylene and added under nitrogen to the previous mixture. Then, 110 mL (1 mmol) of chlorodimethylsilane was added dropwise and an initial sample was removed. The yellow transparent solution was stirred at 20°C for 2 h, then at 35°C for 12 h. Samples were removed after 30 min and 2 h of reaction at 20 °C and after 12 h at 35°C. The disappearance of the vinylic protons signal, from 6 to 5.8 ppm and from 5.1 to 5 ppm, was clearly observed by <sup>1</sup>H NMR after 12 h at 35 °C, indicating that the hydrosilation reaction was successful.

#### 11A2. Reduction of the silicon-chloride bond

The excess of chlorodimethylsilane was evaporated from the previous mixture under reduced pressure for 30 min. Then this mixture was added dropwise to a solution of AlLiH<sub>4</sub> (40 mg; 1 mmol) in 10 mL of anhydrous ether, under nitrogen. The reaction mixture was stirred at 50°C under reflux for 12 h.

#### 11A3. Hydrosilation of allyl-2-bromoisobutyrate by Si-H terminated PP

Concentration ratio: *olefin* : *silane* : *catalyst* : *ligand* = 5 : 1 : 0.025 : 0.05

The reaction was conducted in xylene. 300-mg of silane-terminated PP (0.1 mmol) was dissolved into 3 mL of xylene, and placed into a 25 mL three-neck round bottom flask under nitrogen. 1.04 mg (6 mmol) of methylnaphtoquinone was then added, followed by 31.1 mL (0.096 mol/L; 3 mmol) of Karstedt's catalyst in xylene. Then, 104 mg (0.5 mmol) of allyl-2-bromoisobutyrate was added at once. An initial sample was removed and the mixture was stirred under nitrogen at 30°C for 12 h, then at 50°C for 8 h and finally at 80°C for 30 h. Aliquots were removed after 2 h and 12 h at 30°C, 8 h at 50°C and 30 h at 80°C. The polymer was precipitated into 100 mL of methanol at -15°C for 12 h.

The <sup>1</sup>H NMR spectra revealed the absence of the expected triplet at 4.15 ppm, assigned to the methylene group linked to the oxygen. This fact seems

to indicate that either this hydrosilation step or the previous reduction step was unsuccessful.

*11A5. Synthesis of allyl-2-bromoisobutyrate*

A 2 L three-neck round bottom flask, previously dried in the oven for 2 hours, was charged with 145.5 mL (1.176 mol) of 2-bromoisobutyrylbromide in 1200 mL of anhydrous ether, under nitrogen. The flask was put into an ice bath and a mixture of triethylamine (204.9 mL; 1.47 mol) and allyl alcohol (40 mL; 0.588 mol) was added dropwise, under nitrogen. The white heterogeneous mixture was constantly stirred under nitrogen with a mechanical stirrer for 7 h at 0°C. The mixture was then filtered twice and the liquid phase was quenched with 600 mL of distilled water. The ether phase was washed four times with 250 mL portions of aqueous NaOH (5 %), twice with 250 mL portions of aqueous HCl (5%), and twice more with 250 mL portions of distilled water. The solution was then dried over magnesium sulfate for 1 hour and magnesium sulfate was filtered off. Ether was evaporated under reduced pressure and the product was distilled under vacuum (bp=65°C under 5 mmHg) to afford 84 g of a colorless liquid (yield 69 %). The purity of the final product was totally confirmed by IR, but only partially by <sup>1</sup>H NMR. So it has to be further established by <sup>13</sup>C NMR.

*11B. Synthesis of a Polypropylene Macroinitiator (Route2).*

*11B1. Hydrosilation of allyl-2-bromoisobutyrate by tetramethyldisiloxane*

Concentration ratio: *olefin : silane : catalyst : ligand* = 1 : 14 : 0.02 : 0.04

The reaction was conducted in bulk. A 250 mL three-neck round bottom flask was backfilled with nitrogen and charged with tetramethyldisiloxane (100 mL; 565.8 mmol) and allyl-2-bromoisobutyrate (8.316 g; 40 mmol). The flask was then put into an ice bath and a mixture of Karstedt's catalyst (0.096 mol/L; 8.32 mL; 0.8 mmol) and methylnaphtoquinone (0.274 g; 1.6 mmol) in xylene solution was then added dropwise, leading to a gold homogeneous solution. The mixture was stirred under nitrogen at 20°C for 2 h, then at 50°C for 16 h. The excess of tetramethyldisiloxane was evaporated at 50°C under reduced pressure (50 mmHg then 5 mmHg).

Aliquots were removed after 30, 60 and 120 minutes of reaction at room temperature and after 2 h and 16 h at 50°C.

<sup>1</sup>H NMR spectra show that the reaction was fully completed after 2 hours at 50°C.

5 **11B2. Hydrosilation of vinyl-terminated polypropylene with unpurified 11B1**

Concentration ratio: *vinyl-terminated PP* : *silane* : *catalyst* : *ligand* = 1 : 5 : 0.02 : 0.04 A 50 mL three-neck round bottom flask was backfilled with nitrogen and charged with vinyl-terminated polypropylene (1.5 g; 0.5 mmol). The polymer was dissolved into 3 mL of xylene and tetramethyldisiloxanepropyl-2-

10 bromoisobutyrate (7.71 g; purity ~50%; ~2.5 mmol. A mixture of Karstedt's catalyst (0.096 mol/L; 104.16 mL; 0.01 mmol) and methylnaphtoquinone (3.45 mg; 0.02 mmol) in xylene solution was then added dropwise, leading to an orange homogeneous solution. An initial sample was removed, and the mixture was stirred under nitrogen at 20°C for 1 h, then at 50°C for 2 h. Aliquots were  
15 removed after 30 and 60 minutes of reaction at room temperature and after 2 h at 50°C.

The polymer was precipitated into 50 mL C, re-dissolved into 10 mL of anhydrous ether and filtrated through a 0.2-mm filtration membrane. Ether was evaporated under reduced pressure and a sample was removed for analysis. The  
20 polymer was dissolved again in 10 mL of anhydrous ether and precipitated into 100 mL of methanol. The solvents were then evaporated under low ( $10^{-2}$  mmHg) to high vacuum ( $10^{-7}$  mmHg) for 2 h and 4 h respectively. Another sample was removed for analysis. The disappearance of the vinylic protons signal, from 6 to 5.8 ppm and from 5.1 to 5 ppm, was clearly observed by <sup>1</sup>H NMR after 2 h at 50  
25 °C, indicating that the hydrosilation reaction was successful. In addition, the triplet at 4.15 ppm, assigned to the methylene group linked to the oxygen of the ester, can still be observed after all the purification steps. However, the integrations in the 1-0 ppm region are still higher than expected, indicating that the polymer blend still contain some side products, possibly attached to the polypropylene.

30 Elemental analysis indicates a content of bromine equal to 1.94 % (2.6 % expected, assuming that the molecular weight of the polypropylene is around 3000 g/mol).



ATRP can thus be carried out with this macro-initiator

11B3. *ATRP of MMA using polypropylene-based macro-initiator (preparation of a block copolymer)*

Concentration ratios: monomer : macro-initiator : catalyst : ligand = 100 : 1 :

5 1 : 1

The polymerization was conducted in xylene (solvent : monomer = 1 : 1). The quantity of macro-initiator in the polypropylene blend was assessed from the content of bromine found by elemental analysis (1.94 %). The polypropylene-based macro-initiator (11B2 – 0.46 g ; 1.94 % of Br; 0.112 mmol) in 1.2 mL of  
10 xylene and 1.195 mL (11.2 mmol) of MMA were placed into a sealed Schlenk flask and were cleaned of oxygen by a bubbling of nitrogen for 30 min. The mixture was frozen into a liquid nitrogen bath and the Schlenk flask was degassed and backfilled with liquid nitrogen. Then, copper(I) chloride (11.06 mg; 0.112 mmol) and dNBipy (45.66 mg; 0.112 mmol) were added to the frozen mixture and the  
15 flask was degassed and backfilled with liquid nitrogen 3 times. The liquid nitrogen bath was removed and the mixture was stirred, as it became a dark red liquid. An initial sample was removed and the mixture was stirred at 90 °C for 6 h. Aliquots removed during the polymerization after 30 min, 1 h, 2 h, 4 h and 6 h of reaction. The dark viscous mixture obtained was dissolved into 10 mL of THF and  
20 precipitated into 100 mL of a MeOH/H<sub>2</sub>O solution (80/20) for 12 h at 0 °C. The mixture was then filtrated, leaving a white powder that was dried under reduced pressure for 4 h.

A nice shift toward the high molecular weight was observed, as well as the progressive disappearance of the PP-based macro-initiator. Final  
25 conversion can be assessed from the NMR spectrum of the purified polymer blend. Final monomer conversion value is 61 %, which lead to a macro-initiator efficiency of 32 %. Moreover, the polydispersity index, obtained by cutting the tail of the GPC trace coming from the unreacted macro-initiator, is below 1.1.

In these examples it is believed that the first step involving  
30 tetramethyldisiloxane and allyl-2-bromoisobutyrate leads to the desired product with a significant amount of side products. In addition, the second step affords only partially functionalized polypropylene. Hence, first step was performed again,

at a lower temperature. In addition, evaporation of the excess of tetramethyldisiloxane was conducted while the product was still under nitrogen (rather than transferring the mixture into a round-bottom flask to use the rotavapor) and at room temperature, as formation of Si-O-Si based products can be observed with Karstedt's catalyst in the presence of water; Britcherm L.G., Kehoe, D.C.; Matisons, J.G., Swinser, A.G. "Siloxane Coupling Agents" *Macromolecules* 1995, 28, 3110-3118. Furthermore, a much longer reaction time was allowed for the second step, in order to try to increase the degree of functionality of the polypropylene.

*11B4: Hydrosilation of allyl-2-bromoisobutyrate by tetramethyldisiloxane*

Concentration ratio: *olefin : silane : catalyst : ligand* = 1 : 10 : 0.02 : 0.04

The reaction was conducted in bulk. A 100 mL three-neck round bottom flask, previously dried in the oven for 2 h, was backfilled with nitrogen and charged with tetramethyldisiloxane (42.7 mL; 241.5 mmol) and allyl-2-bromoisobutyrate (5 g; 24.15 mmol). The flask was then put into an ice bath and a mixture of Karstedt's catalyst (0.096 mol/L; 5.03 mL; 0.483 mmol) and methylnaphthoquinone (0.266 g; 0.966 mmol) in xylene was then added dropwise, leading to a gold homogeneous solution. The mixture was stirred under nitrogen at 30°C for 2 h, then at 50°C for 40 h. The excess of tetramethyldisiloxane and the possible residual allyl-2-bromoisobutyrate were evaporated under reduced pressure (5 mmHg) and high vacuum ( $10^{-7}$  mmHg). Aliquots were removed after 1 h and 2 h of reaction at 30°C and after 2, 16 and 40 h at 50°C.  $^1\text{H}$  NMR spectra show the complete disappearance of the allylic proton signals of the allyl-2-bromoisobutyrate between 6 and 5.85 ppm and between 5.5 and 5.2 ppm after 40 h, indicating that the reaction isn't completed until this rather long time. In addition, the triplet at 4.15 ppm assigned to the methylene group linked to the oxygen of the ester can clearly be observed, with an integration consistent with those of the Si-H and of the 2  $\text{CH}_3$  attached to the C-Br. However, the integrations in the 0.5-0 ppm region are still higher than expected, indicating that a significant amount of side-products is obtained again.

*11B5. Hydrosilation of vinyl-terminated polypropylene with unpurified 11B4*

Concentration ratio: *vinyl-terminated PP* : *silane* : *catalyst* : *ligand* = 1 : 5 : 0.02 : 0.04

The reaction was conducted in xylene. A 50 mL three-neck round bottom flask was backfilled with nitrogen and charged with vinyl-terminated polypropylene (1.5 g; 0.5 mmol). The polymer was dissolved into 3 mL of xylene and tetramethyldisiloxanepropyl-2-bromoisobutyrate (1.71 g; purity ~50%; ~2.5 mmol). A mixture of Karstedt's catalyst (0.096 mol/L; 104.16 mL; 0.01 mmol) and methylnaphthoquinone (3.45 mg; 0.02 mmol) in xylene solution was then added dropwise, leading to an orange homogeneous solution. An initial sample was removed, and the mixture was stirred under nitrogen at 50°C for 24 h. Aliquots were removed after 2 h and 24 h at 50°C. The polymer was precipitated into 50 mL of methanol, re-dissolved into 10 mL of anhydrous ether and filtrated through a 0.2-mm filtration membrane. It was then precipitated into 100 mL of methanol. The solvents were then evaporated under low ( $10^{-2}$  mmHg) and high vacuum ( $10^{-7}$  mmHg). A sample was removed for elemental analysis and NMR. The disappearance of the vinylic protons signal, from 6 to 5.8 ppm and from 5.1 to 5 ppm, was clearly observed by  $^1\text{H}$  NMR after 2 h at 50 °C, indicating that the hydrosilation reaction was successful. In addition, the triplet at 4.15 ppm, assigned to the methylene group linked to the oxygen of the ester, can still be observed after all the purification steps. However, the integrations in the 0.5-0 ppm region are still higher than expected, indicating that the polymer blend still contain some side products, possibly attached to the polypropylene

#### 11C. Variation of Second Route. (Scheme 7)

##### 11C1. Hydrosilation of vinyl terminated polypropylene with tetramethyldisiloxane.

Concentration ratio: *vinyl-terminated PP* : *silane* : *catalyst* : *ligand* = 1 : 5 : 0.02 : 0.04

A 25 mL three-neck round bottom flask, previously dried in the oven for 3 h, was backfilled with nitrogen and charged with 2.07 mg (12 mmol) of methylnaphthoquinone. The Karstedt's catalyst in xylene solution (0.096 mol/L, 62.5 mL, 6 mmol) was then added at once, leading to a gold homogeneous solution. 1.23 g (0.3 mmol) of vinyl-terminated polypropylene was dissolved into 2.5 mL of xylene and added under nitrogen to the previous mixture. Then, 265.2

mL (1.5 mmol) of tetramethyldisiloxane was added at once and an initial sample was removed. The yellow transparent solution was stirred at 35°C for 2 h.

Aliquots were removed after 1 and 2 h of reaction. The disappearance of the vinylic protons signal, from 6 to 5.8 ppm and from 5.1 to 5 ppm, was clearly

5 observed by  $^1\text{H}$  NMR, indicating that the hydrosilation reaction was successful.

The absence of coupling between two PP chains was confirmed by GPC.

*11C2. Hydrosilation of allyl-2-bromoisobutyrate (ABiB) with the previous functionalized polypropylene.*

Concentration ratio: *silane-terminated PP* : *ABiB* : *catalyst* : *ligand* = 1 : 5 :

10 0.02 : 0.04

The excess of tetramethyldisiloxane was evaporated under reduced pressure from the previous mixture (JS 1/19) for 1h. A mixture of Karstedt's catalyst in xylene (6 mmol; 65.5 mL) and methylnaphtoquinone (12 mmol; 2.07 mg) was then added, followed by 311.8 mg (1.5 mmol) of allyl-2-

15 bromoisobutyrate. An initial sample was removed and the mixture was stirred at 50°C for 12 h. The polymer was precipitated into 100 mL, re-dissolved into 10 mL of anhydrous ether and filtrated through a 0.2-mm filtration membrane. Ether was evaporated under reduced pressure and a sample was removed for analysis. The polymer was dissolved again in 10 mL of anhydrous ether and precipitated into

20 200 mL of methanol. The solvents and the possible excess of allyl-2-bromoisobutyrate were then evaporated under high vacuum ( $10^{-7}$  mmHg) for 5 h. Another sample was removed for analysis. Evidence for efficiency of the reaction is brought by observing the triplet at 4.15 ppm, assigned to the methylene group linked to the oxygen of the ester. However, efficiency of the reaction cannot be  
25 assessed by NMR, due to the weakness of the aforementioned signal. Thus, further ATRP will be based again on the content of bromine found by elemental analysis.

*11C3. ATRP of MMA using Polypropylene-based Macroinitiator*

The quantity of macro-initiator in the polypropylene blend was  
30 assessed from the content of bromine found by elemental analysis (1.16 %) and this was used to calculate the concentration ratios: monomer : macro-initiator : catalyst : ligand = 100 : 1 : 1 : 1. The polypropylene-based macro-initiator (JS

1/20 – 0.671 g; 1.16 % of Br; 97.4 mmol) in 2.1 mL of xylene, 1.042 mL (9.74 mmol) of MMA and 45.5 mL of diphenylether (internal standard, 5 % wt / monomer) were placed into a sealed Schlenk flask and were cleaned of oxygen by a bubbling of nitrogen for 30 min. The mixture was frozen into a liquid nitrogen bath and the Schlenk flask was degassed and backfilled with liquid nitrogen. Then, copper(I) chloride (9.64 mg; 97.4 mmol) and dNBipy (39.82 mg; 97.4 mmol) were added to the frozen mixture and the flask was degassed and backfilled with liquid nitrogen 3 times. The liquid nitrogen bath was removed and the mixture was stirred, as it became a dark red liquid. An initial sample was removed and the mixture was stirred at 90°C for 30 h. Aliquots removed during the polymerization after 30 min, 1 h, 2 h, 4 h, 6 h, 8 h, 20 h and 30 h of reaction. The dark viscous mixture obtained was dissolved into 5 mL of THF and precipitated into 100 mL of MeOH for 12 h at -15°C. The mixture was then filtrated, leaving a white powder that was dried under high vacuum ( $10^{-7}$  mmHg) for 4 h. A nice shift toward the high molecular weights was observed, as well as the progressive disappearance of the PP-based macro-initiator. However, the final molecular weight of the copolymer was significantly higher than expected (27,000 g/mol rather than 15,000 g/mol expected). This indicates either a very slow initiation process or poor macro-initiator efficiency which can be attributed to the lower degree of functionality of the starting polypropylene (1.16 % bromine versus 1.94 % previously). Thanks to a much better homogeneity of the mixture, the final monomer conversion obtained is significantly higher, around 80 % (61 % previously).

### 13. *ATRP of MMA using macroinitiator and ionic liquids.*

In all experiments describing the polymerization of MMA in ionic liquids using low molecular weight alkyl halide (EBiB) as initiator; co-pending U.S. Application 60/238,809, low initiation efficiencies were observed. Higher initiation efficiencies were observed when the ionic liquid was used in catalytic amounts. The concentration of radicals in ATRP equilibrium is determined by the concentration of the transition metal halides and alkyl halides. The ionic liquids are highly solvating, yet non-coordinating solvents. It is plausible that the concentration of catalyst (both activator and deactivator) is low in organic phase

where polymerization occurs but very high in ionic liquid phase. Thus, when low molecular weight initiator enters ionic liquid phase, it encounters very high catalyst concentration ( $\sim 1$  mole), which results in the generation of very high concentration of radicals leading to termination, i.e. low initiation efficiency.

5 However, this short non-stationary state is over when higher molecular weight oligomers are formed which cannot enter ionic liquid phase. Then well-controlled polymerization occurs with linear evolution of molecular weight with conversion. The entire molecular weight distribution shifts to higher masses and polymers with relatively low polydispersities are obtained. If a macroinitiator is used instead of  
10 ethyl 2-bromoisobutyrate, it should stay in the organic phase and the ATRP process should be determined only by the migration of the catalyst to this phase.

In order to verify this scenario, macroinitiators were used (PMMA-Br with  $M_n = 1500$ ) in the ATRP of MMA with 1-butyl-3-methylimidazolium bromide- $\text{FeBr}_2$ , in the presence of  $\text{Fe}^{\text{III}}$ . The GPC traces of polymer samples taken at  
15 intervals throughout the polymerization show an initial bimodal molecular weight distribution with progressively and continuously occurring initiation as the macroinitiator signal is converted to higher molecular weight polymer. The first-order kinetic plots were linear, showing a constant concentration of radicals throughout the reaction. The reaction rates depend strongly on the amount of the  
20 deactivator in the ionic liquid phase. Thus, when the concentration of  $\text{Fe}^{\text{II}}$  was increased from 0.7 to 0.9 mole vs. initiator, the apparent polymerization rate constant increased from  $k_{\text{app}} \approx 5.1 \cdot 10^{-6} \text{ s}^{-1}$  to  $\approx 2 \cdot 10^{-5} \text{ s}^{-1}$ . Total molecular weight follows closely the theoretical values, while initiation efficiency was high. When only the high molecular weight fraction of the polymer is analyzed, the initiation  
25 efficiency continuously increases. It is possible to determine the rate of disappearance of the macroinitiator in this system, by analyzing number distribution of polymer chains (RI signal divided by molecular weight). [K. Matyjaszewski; D. A. Shipp; G. P. McMurtry; S. G. Gaynor; T. Pakula *J. Polym. Sci., Part A: Polym. Chem.* 38, 2023 (2000).] It appears that half of the  
30 macroinitiator is consumed after 5 hours and 90% after 15 hours. Relatively slow consumption of the macroinitiator is probably due to low concentration of the activator in the organic phase. The number of polymer chains vary linearly in time,

but the increase in rate is much larger than it would be expected for a small increase of catalyst concentration. This observation suggests that not only the activation step but also deactivation, and thus the concentration of metal in higher oxidation state in the organic phase, determine the ATRP equilibrium when ionic liquids are used. At high degree of polymerization the ATRP equilibrium is altered not only because of an increase in viscosity, but because the diffusion of the catalyst, solvated by the ionic liquid, into the organic phase is decreased. ATRP in ionic liquids proceeds with low initiation efficiency. This can be ascribed to relatively low concentration of the catalyst in organic medium and very high concentration of catalyst in ionic liquid phase. When low molecular weight initiator enters ionic liquid phase, it generates large concentration of radicals, which terminate and result in low initiator efficiency. However, after this short non-stationary state, polymerization is controlled and molecular weights evolve linearly with conversion, resulting in polymers with low polydispersities. The initiation efficiency can be significantly increased in the presence of macroinitiators, which cannot enter ionic liquid phase.

It is to be understood that the present description illustrates those aspects of the invention relevant to a clear understanding of the invention. Certain aspects of the invention that would be apparent to those of ordinary skill in the art and that, therefore, would not facilitate a better understanding of the invention have not been presented in order to simplify the present description. Although the present invention has been described in connection with certain embodiments, those of ordinary skill in the art will, upon considering the foregoing description, recognize that many modifications and variations of the invention may be employed. All such variations and modifications of the invention are intended to be covered by the foregoing description and the following claims.

## CLAIMS

1. A polymerization process, comprising:  
polymerizing macromonomers with a compatible macroinitiator to form a  
graft (co)polymer.
- 5 2. The process of claim 1, wherein the polymerization process is a radical  
polymerization process.
3. The process of claim 1, wherein the polymerization process is a controlled  
polymerization process.
4. The process of claim 3, wherein the polymerization process is a controlled  
10 radical polymerization process.
5. The process of claim 3, wherein the polymerization process is a controlled  
addition polymerization process.
6. The process of claim 1, wherein the macromonomer comprises silicon.
7. A polymerization process, comprising:  
15 copolymerizing copolymerizable monomers and first (co)polymerizable  
macromonomers with a compatible macroinitiator.
8. The polymerization process of claim 7, wherein the first (co)polymerizable  
monomers and macromonomers are radically copolymerizable.
9. The polymerization process of claim 7 wherein the first (co)polymerizable  
20 macromonomer comprises silicon.
10. The polymerization process of claim 7, wherein the first (co)polymerizable  
macromonomer comprises a biocompatible monomer unit.
11. The polymerization process of claim 7, wherein the first (co)polymerizable  
macromonomer is a polyolefin.



12. The polymerization process of claim 7, wherein the copolymerizable monomers are small molecule monomers.
13. The polymerization process of claim 7, wherein the first macromonomer comprises reactive terminal functionality which form a gradient copolymer with the copolymerizable monomer.
14. The polymerization process of claim 13, further comprising:  
polymerizing a second copolymerizable macromonomer comprising reactive terminal functionality different than the reactive functionality on the first copolymerizable macromonomer.
15. The polymerization process of claim 14, further comprising:  
forming a graft copolymer with a homogeneous distribution of grafts.
16. The polymerization process of claim 14, wherein a ratio of the reactivity of the copolymerizable macromonomers to the reactivity of the copolymerizable monomer is within the range of 0.5 to 1.5.
17. The polymerization process of claim 14, wherein the first and second copolymerizable macromonomers comprise different monomer units.
18. The process of claim 17, further comprising:  
forming a gradient copolymer.
19. The process of claim 15, wherein the grafts comprise at least one lactic acid unit and the molecular weight distribution of the backbone is less than 2.
20. The process of claim 17, wherein the macroinitiator comprises a gradient or a block copolymer segment.
21. A polymerization process, comprising:  
polymerizing free radically polymerizable monomers and free radically polymerizable macromonomers with a macroinitiator, wherein in the macroinitiator comprises a graft copolymer.

22. The process of claim 21, further comprising:  
forming a graft copolymer comprising cross linking functional groups; and  
crosslinking the copolymer to stabilize the morphology of the bulk graft  
copolymer.
- 5 23. A graft copolymer, comprising:  
a backbone comprising free radically polymerizable monomer units; and  
graft (co)polymer segments distributed along the backbone.
24. The graft copolymer of claim 23, wherein the graft (co)polymer segments  
are distributed uniformly along the backbone.
- 10 25. The graft copolymer of claim 23, wherein the graft (co)polymer segments  
are distributed along the backbone with a higher concentration of the graft  
(co)polymer segments at one end of the backbone.
26. The graft copolymer of claim 23, wherein the graft (co)polymer segments  
are distributed along the backbone with a higher concentration of the graft  
15 (co)polymer segments at both ends of the backbone.
27. The graft copolymer of claim 23, wherein the graft (co)polymer segments  
are block copolymers and each block comprises different monomer units.
28. The graft copolymer of claim 27, wherein the copolymer forms a single  
homogeneous phase.
- 20 29. The graft copolymer of claim 27, wherein the copolymer forms a biphasic  
copolymer.
30. The graft copolymer of claim 27, wherein the copolymer forms a triphasic  
copolymer.
31. The graft copolymer of claim 27, further comprising:  
25 reactive functional groups.

32. The graft copolymer of claim 31, wherein the reactive functional groups are capable of stabilizing a morphology of the graft copolymer.

33. The graft copolymer of claim 32, wherein the reactive functional groups are crosslinkable functional groups.

5 34. The graft copolymer of claim 31, wherein the reactive functional groups are capable of crosslinking the one or more phases.

35. The graft copolymer of claim 27, wherein the relative mole fractions of the radically copolymerizable monomer units and graft (co)polymer segments effect the morphology of the graft copolymer.

10 36. The graft copolymer of claim 35, wherein the graft copolymer has a substantially cylindrical morphology for at least one of the phases.

37. The graft copolymer of claim 35, wherein at least two phases have a continuous morphology.



Homogeneously  
distributed  
branches

Polymer A



Heterogeneously  
distributed  
branches

Polymer B

FIGURE 1

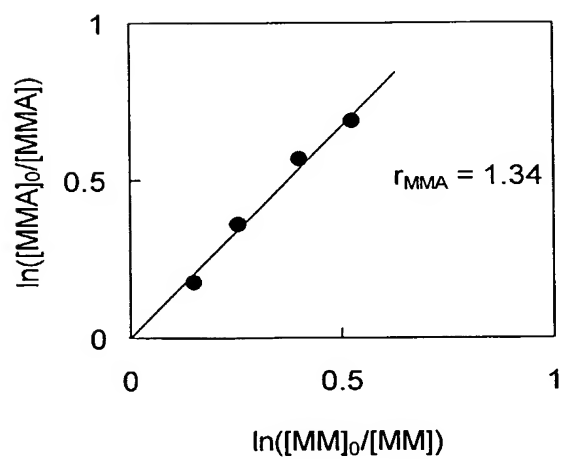
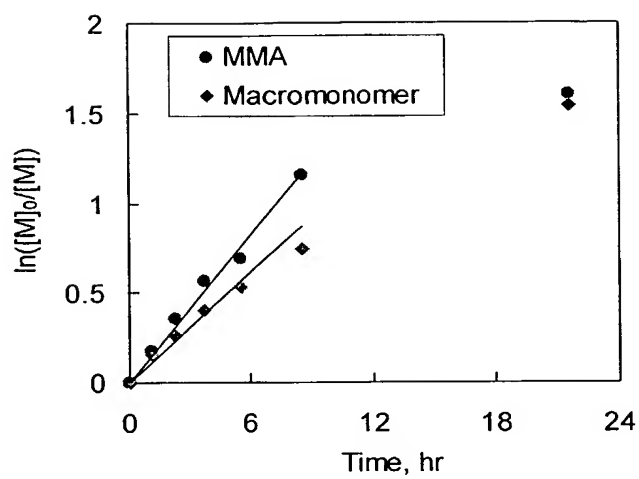
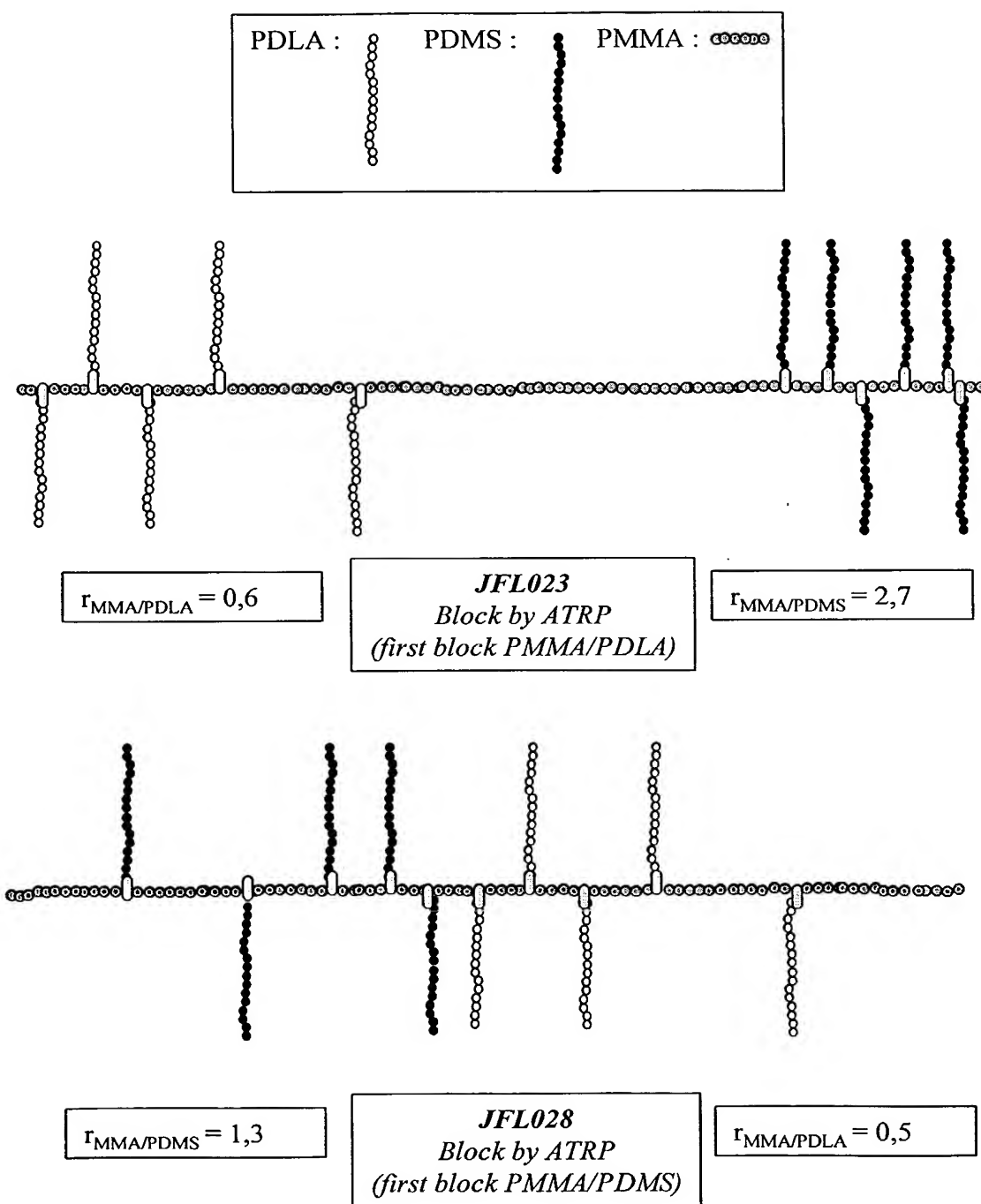
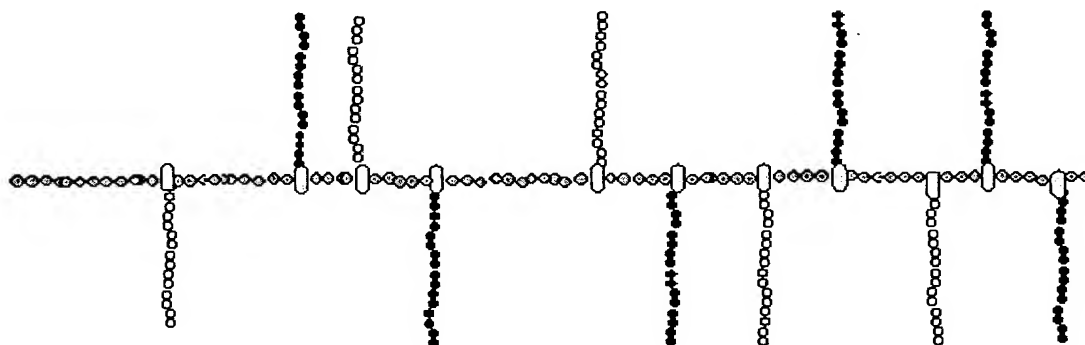


FIGURE 2

## Block copolymers

**FIGURE 3**

**FIGURE 4**

## INTERNATIONAL SEARCH REPORT

Ir      national Application No

PCT/US 01/49523

**A. CLASSIFICATION OF SUBJECT MATTER**

IPC 7    C08F290/02    C08F299/00    C08F293/00

According to International Patent Classification (IPC) or to both national classification and IPC

**B. FIELDS SEARCHED**

Minimum documentation searched (classification system followed by classification symbols)

IPC 7    C08F

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practical, search terms used)

EPO-Internal, WPI Data

**C. DOCUMENTS CONSIDERED TO BE RELEVANT**

Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	WO 00 56795 A (UNIV CARNEGIE MELLON) 28 September 2000 (2000-09-28) claims 242-247 ---	1-6
X	EP 0 870 809 A (FUJI PHOTO FILM CO LTD) 14 October 1998 (1998-10-14) column 7, line 6-13; examples ---	23,24
A	US 5 789 487 A (MATYJASZEWSKI KRZYSZTOF ET AL) 4 August 1998 (1998-08-04) the whole document ---	1
A	US 5 626 863 A (HUBBELL JEFFREY A ET AL) 6 May 1997 (1997-05-06) claim 1 -----	10,23

☐ Further documents are listed in the continuation of box C.☒ Patent family members are listed in annex.

## \* Special categories of cited documents :

- \*A\* document defining the general state of the art which is not considered to be of particular relevance
- \*E\* earlier document but published on or after the international filing date
- \*L\* document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)
- \*O\* document referring to an oral disclosure, use, exhibition or other means
- \*P\* document published prior to the international filing date but later than the priority date claimed

- \*T\* later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
- \*X\* document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone
- \*Y\* document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.
- \* & \* document member of the same patent family

Date of the actual completion of the international search

25 July 2002

Date of mailing of the international search report

06/08/2002

Name and mailing address of the ISA

European Patent Office, P.B. 5818 Patentlaan 2  
 NL - 2280 HV Rijswijk  
 Tel. (+31-70) 340-2040, Tx. 31 651 epo nl,  
 Fax: (+31-70) 340-3016

Authorized officer

Meulemans, R



## INTERNATIONAL SEARCH REPORT

In 1st Application No

PCT/US 01/49523

Patent document cited in search report		Publication date	Patent family member(s)	Publication date
WO 0056795	A	28-09-2000	AU 3919500 A	09-10-2000
			EP 1171496 A1	16-01-2002
			WO 0056795 A1	28-09-2000
<hr/>				
EP 0870809	A	14-10-1998	JP 10339949 A	22-12-1998
			EP 0870809 A2	14-10-1998
			US 6194480 B1	27-02-2001
			US 2001002415 A1	31-05-2001
<hr/>				
US 5789487	A	04-08-1998	AU 3585997 A	02-02-1998
			BR 9710273 A	10-08-1999
			CA 2259995 A1	15-01-1998
			CN 1228789 A	15-09-1999
			EP 0914352 A1	12-05-1999
			JP 2000514479 T	31-10-2000
			TW 397852 B	11-07-2000
			WO 9801480 A1	15-01-1998
			US 6111022 A	29-08-2000
			US 6124411 A	26-09-2000
			US 6162882 A	19-12-2000
			US 5945491 A	31-08-1999
<hr/>				
US 5626863	A	06-05-1997	US 5410016 A	25-04-1995
			US 2002091229 A1	11-07-2002
			US 6060582 A	09-05-2000
			US 5567435 A	22-10-1996
			US 5749968 A	12-05-1998
			US 6306922 B1	23-10-2001
			US 5986043 A	16-11-1999
			US 5468505 A	21-11-1995
			AT 197125 T	15-11-2000
			AU 673160 B2	31-10-1996
			AU 3735393 A	05-10-1993
			AU 683209 B2	06-11-1997
			AU 3780993 A	13-09-1993
			BR 9306038 A	13-01-1998
			BR 9306041 A	18-11-1997
			CA 2117584 A1	02-09-1993
			CA 2117588 A1	16-09-1993
			DE 69329594 D1	30-11-2000
			DE 69329594 T2	31-05-2001
			DK 627911 T3	20-11-2000
			EP 0627911 A1	14-12-1994
			EP 0627912 A1	14-12-1994
			ES 2153378 T3	01-03-2001
			JP 3011767 B2	21-02-2000
			JP 7506961 T	03-08-1995
			JP 3011768 B2	21-02-2000
			JP 7507056 T	03-08-1995
			KR 266912 B1	01-12-2000
			NZ 249770 A	25-09-1996
			NZ 251039 A	26-03-1996
			PT 627911 T	30-04-2001
			WO 9317669 A1	16-09-1993
			WO 9316687 A1	02-09-1993
			US 5843743 A	01-12-1998
			US 5801033 A	01-09-1998
			US 6258870 B1	10-07-2001

PCI/US 01/49523

Form PCT/ISA/210 (patent family annex) (July 1992)

DERWENT-ACC-NO: 2002-675008

DERWENT-WEEK: 200635

***COPYRIGHT 2010 DERWENT INFORMATION LTD***

**TITLE:** Polymerization for preparation of block-graft copolymers, involves polymerizing macromonomers with compatible macroinitiator

**INVENTOR:** LUTZ J; LUTZ J F ; MATYJASZEWSKI K ; SHINODA H

**PATENT-ASSIGNEE:** LUTZ J[LUTZI] , MATYJASZEWSKI K[MATYI] , SHINODA H[SHINI] , UNIV CARNEGIE MELLON [UYCAN]

**PRIORITY-DATA:** 2000US-257738P (December 22, 2000) , 1999US-369157 (August 6, 1999) , 2000US-534827 (March 23, 2000) , 2001US-034908 (December 21, 2001)

**PATENT-FAMILY:**

PUB-NO	PUB-DATE	LANGUAGE
WO 02068485 A1	September 6, 2002	EN
US 20020183473 A1	December 5, 2002	EN
AU 2002231156 A1	September 12, 2002	EN
US 7049373 B2	May 23, 2006	EN

**DESIGNATED-STATES:** AE AG AL AM AT AU AZ BA BB BG BR BY BZ CA CH CN CO CR CU CZ DE DK DM DZ EC EE ES FI GB GD GE GH GM HR HU ID IL IN IS JP KE KG KP KR KZ LC LK LR LS LT LU LV MA MD MG MK MN MW MX MZ NO NZ PL PT RO RU SD SE SG SI S K SL TJ TM TR TT TZ UA UG UZ VN YU ZA ZW AT BE CH CY DE DK EA ES FI FR GB GH GM GR IE IT KE LS LU MC MW MZ NL OA PT SD SE SL SZ TR TZ UG ZM ZW

**APPLICATION-DATA:**



PUB-NO	APPL-DESCRIPTOR	APPL-NO	APPL-DATE
WO2002068485A1	N/A	2001WO- US49523	December 21, 2001
US20020183473A1	N/A	2001US- 034908	December 21, 2001
US 7049373B2	N/A	2001US- 034908	December 21, 2001
AU2002231156A1	Based on	2002AU- 231156	December 21, 2001

**INT-CL-CURRENT:**

TYPE	IPC DATE
CIPP	C08F210/00 20060101
CIPS	C08F255/00 20060101
CIPS	C08F257/00 20060101
CIPS	C08F290/02 20060101
CIPS	C08F290/04 20060101
CIPS	C08F293/00 20060101
CIPS	C08F299/00 20060101

**RELATED-ACC-NO:** 1996-455290 1997-297772 1998-506695 2001-024364

**ABSTRACTED-PUB-NO:** WO 02068485 A1

**BASIC-ABSTRACT:**

**NOVELTY** - Macromonomers are polymerized with a compatible macroinitiator to form a graft (co)polymer.

**DESCRIPTION** - An INDEPENDENT CLAIM is included for graft copolymer which contains a backbone comprising free radically polymerizable monomer units and graft (co)polymer segments distributed along the backbone.

**USE** - For preparation of graft copolymers with a broad range of structures



and compositions including phase separable compositions and compositions with controlled distribution of graft copolymers along the backbone polymer and graft copolymer comprising biologically compatible grafts or grafts comprising polyolefins.

**ADVANTAGE** - The method allows separation of well-defined graft (co) polymers with either homogeneously or heterogeneously distributed branches. The distribution of the branches is primarily controlled by the relative reactivity ratio of macromonomer and the monomer, and the compatibilizing effect of the macroinitiator. The use of macroinitiator accelerates the incorporation of macromonomer into the copolymer in the case of less controlled radical polymerizations in addition to fully controlled radical polymerizations. The effect of use of macroinitiator is evident, when the polymerization is done at higher macromonomer concentration because the low molecular weight growing copolymer chain and macromonomer remain compatible. The use of compatible macroinitiator to control the efficiency of initiation of copolymerization also operates in inherently biphasic systems. The macroinitiator does not migrate to the ionic liquid phase and high initiator efficiency is sealed when the catalyst diffuses to the polymerization phase. This allows the use environmentally benign ionic liquids in place of organic solvents in the preparation of graft copolymers and thereby provides a polymerization process that has lower environmental impact.

**DESCRIPTION OF DRAWING(S)** - The figure shows graphical depiction of graft copolymers.

**EQUIVALENT-ABSTRACTS:**

## POLYMERS

**Preferred Process:** Alternatively, copolymerizable monomers and (co) polymerizable macromonomers (I) are copolymerized with a compatible macroinitiator. A copolymerizable macromonomer (II) comprising reactive terminal functionality different from that of (co)polymerizable macromonomer is polymerized, and a graft copolymer with homogeneous distribution of grafts is formed. The ratio of reactivity of (co)polymerizable macromonomers to that of copolymerizable monomer is 0.5-1.5.

Alternatively, free radically polymerizable monomer and free radically polymerizable macromonomers are polymerized with a macroinitiator comprising a graft copolymer. A graft copolymer comprising cross-linking functional groups is formed, and the copolymer is cross-linked to stabilize the morphology of bulk graft copolymer. The polymerization is a controlled radical or addition polymerization process. The copolymerizable monomers



and (co)polymerizable macromonomers are radically (co)polymerizable.

**Preferred Compounds:** The macromonomer and (co)polymerizable macromonomer (I) comprise silicon. The (co)polymerizable macromonomer (I) comprises a biocompatible monomer unit, preferably a polyolefin. The copolymerizable monomers are small molecule monomers. The (co)polymerizable macromonomer (I) comprises a reactive terminal functionality which forms a gradient copolymer with the copolymerizable monomer. The (co)polymerizable macromonomers (I) and (II) comprise different monomer units, and form a gradient copolymer. The graft comprises at least one lactic acid unit and the molecular weight distribution of the backbone is less than 2.

**Preferred Copolymer:** The graft (co)polymer segments in the graft copolymer are distributed uniformly along the backbone with a higher concentration of graft (co)polymer segments at one or both ends of the backbone. The graft (co)polymer segments are block copolymers, and each block comprises different monomer units. The copolymer forms a single homogeneous phase, a biphasic or triphasic copolymer. The graft copolymer contains reactive functional groups which are capable of stabilizing morphology of graft copolymer. The reactive functional groups are cross-linkable functional groups. The relative mole fractions of the radically copolymerizable monomer units and graft (co)polymer segments effect the morphology of the graft copolymer. The graft copolymer has substantially cylindrical morphology for at least one of the phases. At least two phases have a continuous morphology.

Polydimethyl siloxane (PDMS) macroinitiator containing 2-bromoisobutyrate terminal group was synthesized by anionic polymerization of D3 terminated by chlorodimethyl silane, followed by reaction with 3-butenyl-2-bromoisobutyrate. The yield was 78%. A Schlenk flask containing a PDMS-containing macromonomer having number average molecular weight ( $M_n$ ) of 2200 and polydispersity index (PDI) of 1.18 and a PDMS-macromonomer having  $M_n$  of 15600 and PDI of 1.10, was evacuated overnight and back filled with nitrogen. In a 25 ml round bottom flask, 2.5 mg copper chloride and 20.7 mg 4,4'-di(n-nonyl)2,2'-bipyridine, were taken. The flask was capped with a rubber septum and purged with nitrogen gas for half an hour. Then, 0.72 g of deoxygenated methylmethacrylate and 1.62 ml of deoxygenated xylene were added and the mixture was stirred at room temperature under nitrogen for half an hour until a homogeneous maroon solution formed. The solution was cannula-transferred to the Schlenk flask with a nitrogen flow. Polymerization was performed at 90degreesC and periodic sampling and measurements were done. GPC traces showed consistent incorporation of macromonomer into the formed graft



copolymer. The reactivity ratio of methylmethacrylate was found to be 1.37.

CHOSEN-DRAWING: Dwg.1/4

TITLE-TERMS: POLYMERISE PREPARATION BLOCK GRAFT  
COPOLYMER COMPATIBLE

DERWENT-CLASS: A17

CPI-CODES: A02-A; A04-H00H; A10-C02; A10-C03; A10-E01;

ENHANCED-POLYMER-INDEXING: Polymer Index [1.1] 018 ; G0022\*R D01  
D51 D53 H0146; G0384 G0339 G0260  
G0022 D01 D11 D10 D12 D26 D51 D53  
D58 D63 D85 F41 F89 H0146 R00479  
7200; G0102 G0022 D01 D02 D12 D10  
D19 D18 D31 D51 D53 D58 D76 D88  
H0146 R00708 368; G0022\*R D01 D51  
D53 G0419 G0384 G0339 G0260 G0022  
D12 D10 D26 D58 D63 F41 F89 D11 D95  
F91 F81 F86 H0204; H0044\*R H0011;  
H0088 H0011; L9999 L2528 L2506;  
K9723; H0022 H0011; H0033 H0011;  
L9999 L2664 L2506; C999 C088\*R  
C000; M9999 M2153\*R; M9999 M2233  
M2222; M9999 M2324; L9999 L2391;  
L9999 L2153\*R; L9999 L2324; L9999  
L2233 L2222; P1741; P0088;

Polymer Index [1.2] 018 ; ND03; ND04;  
B9999 B4977\*R B4740; B9999 B5005  
B4977 B4740; B9999 B3009; B9999  
B3418\*R B3372; K9756\*R K9745;

Polymer Index [1.3] 018 ; D01 D11 D10  
D50 D63 D95 F91 F41 F81 F86; C999  
C293; C999 C102 C000;

Polymer Index [2.1] 018 ; G0351 G0340  
G0339 G0260 G0022 D01 D11 D10 D12  
D26 D51 D53 D58 D63 D87 F41 F89  
R01130 8433; G0351 G0340 G0339  
G0260 G0022 D01 D11 D10 D12 D26  
D51 D53 D58 D63 D87 F41 F89 R09390



5543; H0191; M9999 M2017; M9999  
M2153\*R; M9999 M2186; M9999 M2813;  
M9999 M2233 M2222; M9999 M2324;  
L9999 L2391; L9999 L2153\*R; L9999  
L2324; L9999 L2233 L2222; C999 C102  
C000; H0000; L9999 L2573 L2506;  
L9999 L2391; L9999 L2153\*R; P0088;

Polymer Index [2.2] 018 ; H0191; M9999  
M2017; M9999 M2153\*R; M9999 M2186;  
M9999 M2813; M9999 M2233 M2222;  
M9999 M2324; L9999 L2391; L9999  
L2153\*R; L9999 L2324; L9999 L2233  
L2222; C999 C102 C000; P1456 P1445  
F81 F86 D01 D11 D50 D82 Si 4A;

Polymer Index [2.3] 018 ; ND01; B9999  
B5094 B4977 B4740; Q9999 Q6917;

Polymer Index [2.4] 018 ; D01 D11 D10  
D50 D60 D85 D69 F27 F26 F89 F41 Br  
7A; H0226;

**SECONDARY-ACC-NO:**

**CPI Secondary Accession Numbers: 2002-190213**